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The determination of molecular structures by vapor phase electron diffraction.

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THE DETERMINATION OF MOLECULAR STRUCTURES BY VAPOR
PHASE ELECTRON DIFFRACTION

I APPARATUS AND METHOD

II THE STRUCTURES OF $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$,
 CH_3GeBr_3 AND CH_3GeF_3

III THE CONFORMATION OF RACEMIC 4,6-DIMETHYL
TRIMETHYLENE SULFITE

By

Frank J. Mustoe

A Dissertation

Submitted to the Faculty of Graduate Studies through the
Department of Chemistry in Partial Fulfillment of the
Requirements for the Degree of Doctor of
Philosophy at the University of
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Windsor, Ontario

1973

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This work is dedicated to my Mother and Father

ABSTRACT

An electron diffraction apparatus has been constructed which allows the molecular structure of vapor phase molecules to be determined with an accuracy of a few thousandths of an angstrom in favorable cases. This instrument has been used to determine the molecular structures of $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3 . The following bond distances (r_g values) were determined: in dimethylgermane, $\text{Ge-C} = 1.953 \pm 0.003$, $\text{Ge-H} = 1.56 \pm 0.04$, $\text{C-H} = 1.074 \pm 0.016\text{\AA}$; in tetramethylgermane, $\text{Ge-C} = 1.955 \pm 0.005$, $\text{C-H} = 1.08 \pm 0.03\text{\AA}$. On correction to an r_e value the Ge-C bond length is in excellent agreement with microwave results for dimethyl germane and the other methyl germanes. This observation parallels the results for the Si-C bond length in the methyl silanes, although in contrast d orbital involvement of germanium is not necessary to explain the Ge-C bond length.

Progressive halogen substitution is found to decrease the Ge-X and Ge-C bond lengths in germanium compounds. The molecular structures of trifluoro(methyl)germane and tribromo(methyl)germane were determined with the following parameters: in CH_3GeF_3 , $\text{Ge-C} = 1.921 \pm 0.012$, $\text{Ge-F} = 1.708 \pm 0.003$, $\text{C-H} = 1.14 \pm 0.08\text{\AA}$, $\angle \text{CGeF} = 114 \pm 0.1^\circ$, $\angle \text{FGeF} = 104.6 \pm 0.9^\circ$; in CH_3GeBr_3 , $\text{Ge-C} = 1.87 \pm 0.04$, $\text{Ge-Br} = 2.284 \pm 0.002\text{\AA}$, $\angle \text{CGeBr} = 111.2 \pm 0.3^\circ$, $\angle \text{BrGeBr} = 107.7 \pm 0.3^\circ$. The Ge-C and Ge-X bond lengths in these molecules have been interpreted

as evidence for a small or negligible amount of d-orbital involvement.

The molecular geometry of the racemic trans-4,6-dimethyl trimethylene sulfite has been studied in the gas phase by electron diffraction. Six trial conformations were examined by least squares fitting of the intensity data. A significance test applied to the fits obtained for each of the six static models tested indicate that the best single representation of the molecule has a chair conformation, although a mixture containing other conformations could not be definitely eliminated. It was not possible to conclusively discriminate between an axial or an equatorial S=O bond, although the axial position is favored. The following bond distances and angles were determined: C-H = 1.119 ± 0.020 , C-C = 1.505 ± 0.014 , C-O = 1.413 ± 0.003 , S-O = 1.480 ± 0.022 , S-O = 1.622 ± 0.009 Å; \angle CCO = 111.1 ± 1.3 , \angle OS=O = 116.7 ± 1.4 , \angle COS = 114.1 ± 1.0 , \angle OSO = $100.7 \pm 2.0^\circ$. The following angles were assumed to be tetrahedral: \angle CCC, \angle HCH, \angle CCH.

All the parameters listed above are based on r_g values with uncertainties three times those calculated from the least squares analysis. The standard errors are believed to include systematic errors in the analysis.

ACKNOWLEDGEMENTS

Every piece of work, no matter how original, is but an extension of the efforts of others. The foundation for this thesis will be found in the references contained therein, but on a more immediate level special thanks are due to a relatively small number of individuals who freely gave of their time and expertise to encourage and direct this research. Dr. A. van Wijngaarden provided many valuable suggestions during construction of the diffraction camera. Compounds for analysis were provided by Drs. G. Wood and M. Miskow (racemic 4,6-dimethyl trimethylene sulfite) and J. Drake and R. Hemmings (the halo(methyl) germanes).

It has been a special privilege to work closely with my research supervisor, Dr. J. L. Hencher, as a colleague and a friend; this work would have been impossible without his understanding and dedication. The cooperation of Dr. S. H. Bauer and his co-workers at Cornell University was invaluable and readily given.

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Chapter 1

INTRODUCTION

The first gas electron diffraction pattern was recorded by Mark and Wierl (1) in 1930, and the following year Wierl (2) published the results of an investigation of the molecular structures of some twenty compounds in which electrons were diffracted by vapor phase molecules. These studies were accomplished by visually estimating optical densities directly from the photographic plates. This subjective procedure was further handicapped by the very rapid decrease in intensity of scattered electrons with increase in scattering angle, which is proportional to about the third power of the sine of the scattering angle. This problem is now overcome experimentally by rotating a sector immediately in front of the photographic plate. The sector reduces the intensity of electrons striking the plate at small angles relative to that at large angles, bringing the density across the whole of the plate within the useful dynamic range of the emulsion. The sector improvement was suggested by Finbak (3) and introduced by Debye (4) in 1939. Optical densities are now obtained from the photographic plates using a microdensitometer.

The diffracted electrons make an interference pattern which can be analysed to deduce information about the molecules causing the scattering. The total experimental intensity I_x which is recorded is a sum of contributions:

$$I_x(s) = \phi(s)[I_m(s) + I_a(s) + I_{ext}(s)] \quad (I.1)$$

where $I_m(s)$ is the desired molecular component (5):

$$I_m(s) = P \sum_{ij} \frac{|f_i(s)| |f_j(s)| \cos(\eta_i - \eta_j) \exp(-\ell_{ij}^2 s^2 / 2) \sin(sr_{ij} - \ell_{ij}^2 s / r_{ij} - a \ell_{ij}^4 s^3 / 6)}{s^4} \quad (I.2)$$

P is a scale factor which is the asymptotic atomic scattering at large angles: $P = \sum_i (Z_i^2 + Z_i)$ where Z_i is the atomic number.

f_i is the form factor for electrons which is related to the X-ray form factor F in the first Born approximation:

$$|f_i(s)| = Z_i - F_i(s)$$

η_i is a phase shift

r_{ij} and ℓ are the interatomic distances and mean square amplitudes of vibration respectively.

a is a parameter used to account for anharmonic molecular motions.

$I_a(s)$ is due to atomic and inelastic scattering:

$$I_a(s) = \frac{1}{s^4} \sum_i (|f_i(s)|^2 + S_i(s)) \quad (I.3)$$

where $S_i(s)$ denotes the inelastic scattering

$I_{ext}(s)$ represents extraneous scattering from parts of the diffraction apparatus where s is the scattering variable

$$s = 4\pi/\lambda \sin(\theta/2) \quad (I.4)$$

λ is the wavelength of the electron beam and θ is the angle of diffraction.

These contributions are modified by the shape of the sector opening ϕ . The last two terms in (I.1) are independent of the molecular structure and may be regarded as experimental background B , which is a smooth curve cleaving the oscillations in I_x .

$$B(s) \doteq I_a(s) + I_{\text{ext}}(s) \quad (\text{I.5})$$

When the sector function is known, the experimental molecular scattering $M_{\text{exp}}(s)$ can be obtained by subtracting the background:

$$M_{\text{exp}}(s) = \frac{s^4 I_x(s)}{I_a(s)\phi(s)} - \frac{s^4 B(s)}{I_a(s)} \quad (\text{I.6})$$

When the sector function is unknown, I_{ext} is assumed to be negligible; then, the experimental molecular component is determined by dividing by the background:

$$M_{\text{exp}}(s) = \frac{I_x(s)}{B(s)} - 1 \quad (\text{I.7})$$

The theoretical expression for the molecular scattering corresponding to $M_{\text{exp}}(s)$ is

$$M_t(s) = \frac{I_m(s)}{I_a(s)} \quad (\text{I.8})$$

$M(s)$ is fit to $M_{\text{exp}}(s)$ by least squares refinements on the molecular

parameters r_g and λ .

A probability function of interatomic distances, known as the radial distribution $f(r)$, can be obtained as the Fourier transform of the nuclear molecular scattering $I'_m(s)$:

$$f(r) = \int_0^{\infty} I'_m(s) \sin s r ds \quad (I.9)$$

The nuclear molecular scattering $I'_m(s)$ is obtained by subtracting the component $\Delta I_m(s)$ due to the electrons from $I_m(s)$:

$$\Delta I_m(s) = I_m(s) - \frac{P}{4\pi} \sum_i \sum_j Z_i Z_j \exp(-\lambda_{ij}^2 s^2 / 2) \sin(Dr_{ij} - \lambda_{ij}^2 s^2 / r_{ij} - a \lambda_{ij}^4 s^3 / 6) \quad (I.10)$$

where $I_m(s)$ is given by equation (I.2).

Since the data do not cover the range $0 < s < \infty$, the integral is replaced by a summation over s_{\min} to s_{\max} with a damping constant b to minimise series termination

$$f(r) = \sum_{s_{\min}}^{s_{\max}} I'_m(s) \sin sr \exp(-bs^2) \Delta s \quad (I.11)$$

where $\exp(-bs_{\max}^2) = 0.1$

Since the first publications by Wierl the electron diffraction technique has become an increasingly useful method for structural analysis. In 1971 Bartell (6) estimated that more than 750 compounds have been studied by vapor phase electron diffraction; at the time of writing this figure must be close to a thousand. Some recent reviews have been written by Bauer (7), Bastiansen, et al (8), Hilderbrandt and Bonham (9), Bartell (6), Karle (10), Kuchitsa (11), Spiridonov (12) and by Hargittai and Lengyel (13).

Chapter II

THE DIFFRACTION RECORDING

An electron diffraction camera has been constructed that enables molecular structure determinations to be made with a precision of a few thousandths of an angstrom in favorable cases. A general view of the camera, high voltage control panel, and gas inlet system is provided in Fig. 1. The diffraction chamber area is shown in more detail in Fig. 2.

The high voltage power supply, gun and electron optics are those of an RCA EMU 3B electron microscope which was rebuilt for the diffraction apparatus. The accelerating voltage was fixed at about 58kV. The beam is defined by a platinum aperture 700 microns in diameter and two further clean up apertures constructed of aluminum, both 1/16 inch in diameter are provided. Fluorescent plates were used to view the beam through a telescope of magnification X30, the diameter of the spot being about 20 microns. The beam current was in the range 10^{-8} to 5×10^{-7} ampere and was measured using a Keithley 414A picoammeter. Exposure times were controlled by a mechanical shutter.

A vacuum of about 1×10^{-5} torr was obtained using a six inch Edwards oil diffusion pump backed by an Edwards oil rotary pump (500.1/s). Scattering patterns were recorded at two camera positions allowing useful data to be obtained over the ranges $s = 2.24 - 15.71\text{\AA}^{-1}$ (camera distance $L = 296.2$ mm) and $s = 11.00 - 47.12\text{\AA}^{-1}$ ($L = 95.5$ mm)

Notations in Figure 1

A General View of the Diffraction Apparatus

- C Diffraction Camera Chamber
- E Electron gun and lens
- H High voltage power supply and lens control panel

SCHEMATIC DIAGRAM

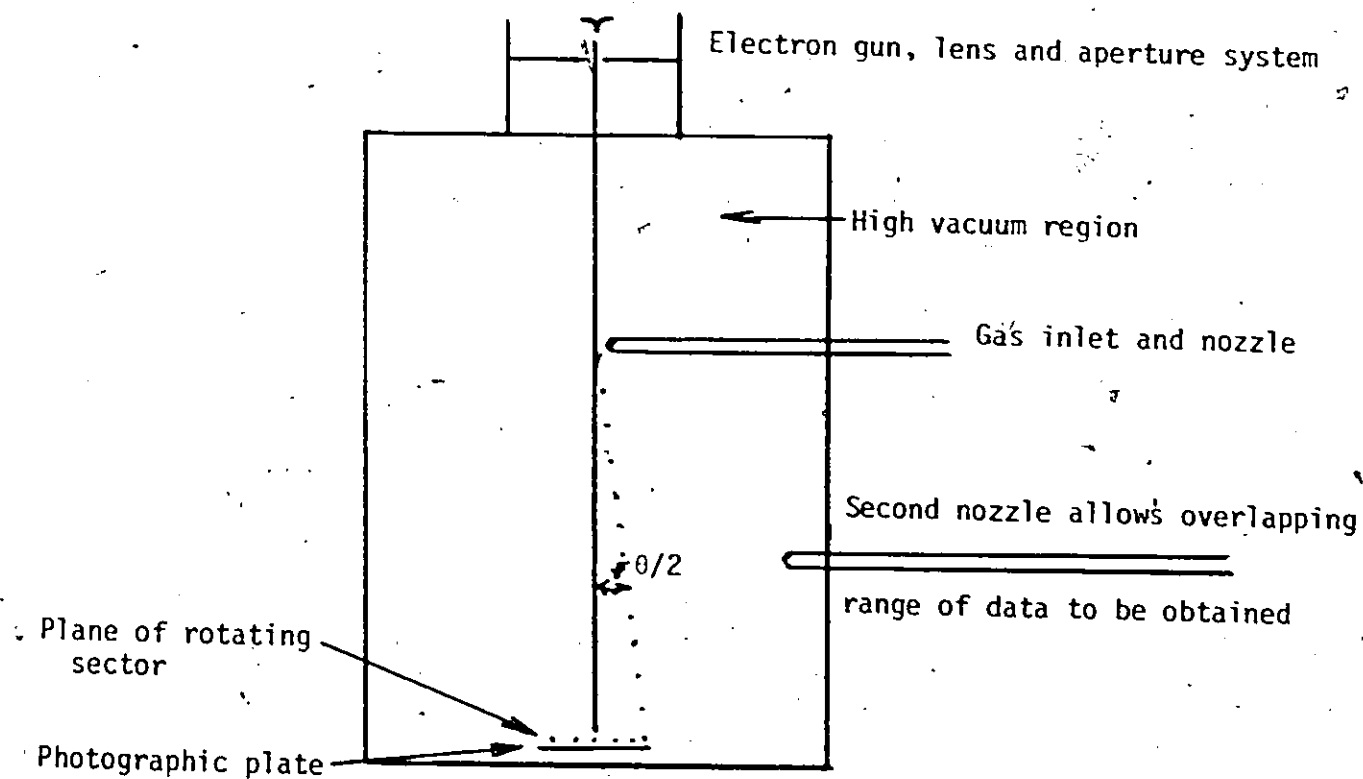
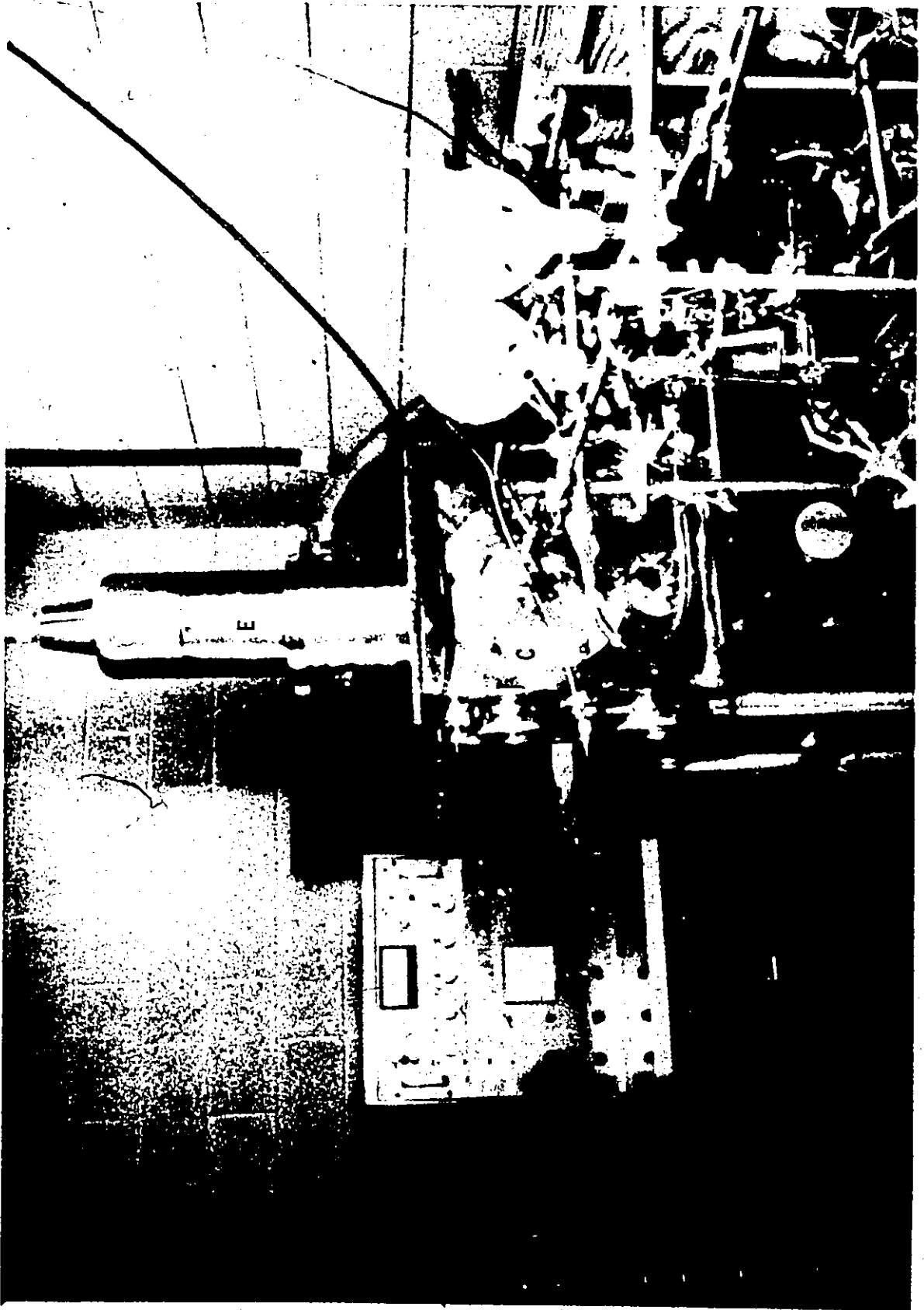


Figure 1

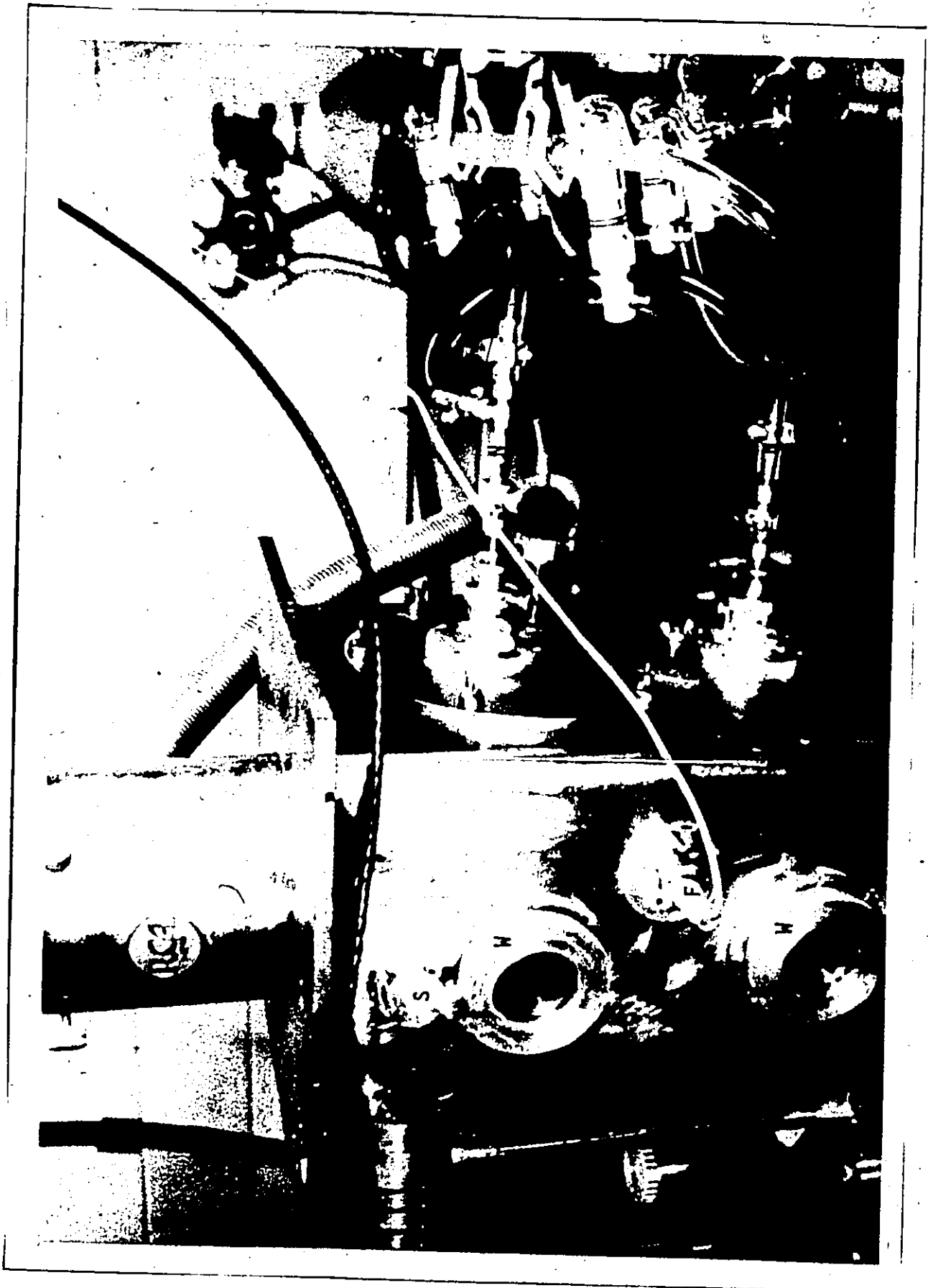


Notations in Figure 2

The Diffraction Chamber

- F Coaxial feedthrough to the Faraday cup
- N Nozzle inlet tubes
- S Mechanical shutter
- W Viewing windows

Figure 2



The nozzle to photographic plate distances L were measured with a cathetometer. The upper nozzle has an exit hole 0.5 mm in diameter, the lower being 0.25 mm diameter. Sample molecules were absorbed on a cryounit covered with activated charcoal and kept at liquid nitrogen temperatures, thus preventing sample delocalisation throughout the chamber. The inside of the chamber is coated with a colloidal suspension of graphite to minimise secondary scattering. The extraneous scattering was checked for each series of patterns by taking a blank exposure. Also included in each series for the purpose of calibrating the scale parameters (L, λ) were an exposure of solid magnesium oxide and another of a vapor sample of carbon disulfide. The MgO sample was offset 11.0 mm from the nozzle, at 90 degrees to the axis of the electron beam, so that it could be rotated into the path of the beam when required.

The Sector

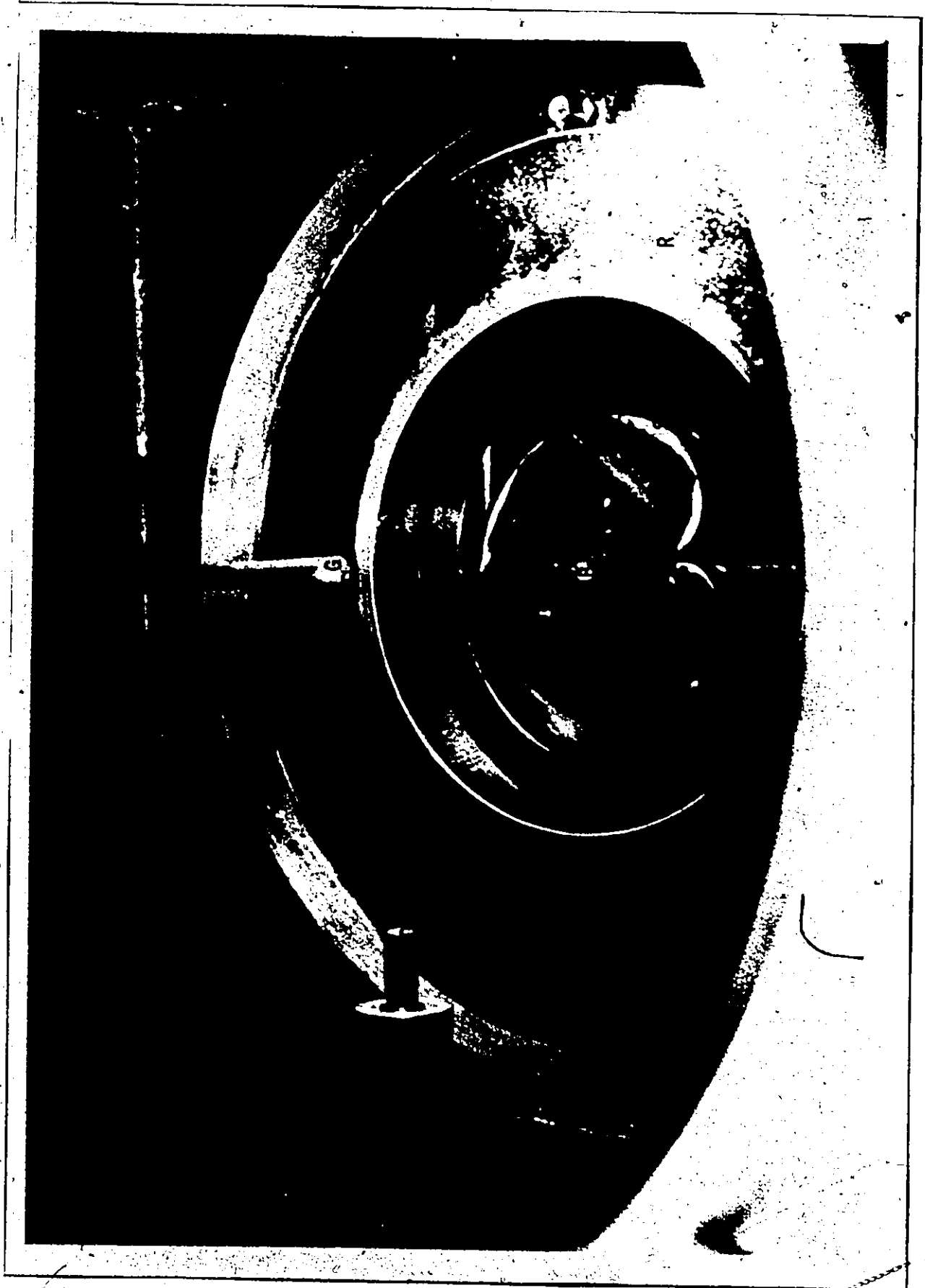
The sector used in this work has an angular opening proportional to the radius cubed, with the function normalised to 175 degrees to allow attachment to the sector race. It is a heart shaped piece of aluminum 1/8 inch thick with a maximum radius of 5.1 cm. The sector race is 15.2 cm in diameter and constructed from 304 stainless steel with bronze ball bearings. The edge of the sector is milled to 45° so that in all cases a very sharp edge was presented to the beam. The sector, which was rotated at about 500 rpm, is shown in Fig. 3 in position 6.25 mm above the photographic plate. A beam stop which operates by centrifugal force is situated at the center

Notations in Figure 3

The Sector Area

- B Beam stop
- D Drive shaft for the sector race. A wheel normally fits over the shaft with the outer edge in contact with the sector race.
- G Gas inlet tube. The nozzle and its attachment have been unscrewed.
- P Photographic plate
- S Sector cut to an r^3 shape.

Figure 3



and consists of a small lead weight attached to a weak spring of phosphor-bronze wire.

A determination of the sector function ϕ is very important for the analysis of diffraction patterns, and was deduced from the experimental scattering by CS_2 using the known molecular structure (14). Starting with the experimental intensity

$$I_x(s) = \phi(s) (I_m(s) + I_a(s) + I_{\text{ext}}(s))$$

and dividing by the background $I_a(s) + I_{\text{ext}}(s)$, the experimental molecular component is obtained as

$$M_{\text{exp}}(s) = \frac{I_x(s)}{\phi(s) (I_a(s) + I_{\text{ext}}(s))} \quad (\text{II.1})$$

This was compared to the theoretical molecular component M_t using an index of resolution R to bring the experimental expression into quantitative agreement with the theoretical:

$$R M_{\text{exp}}(s) = \frac{M_t(s)}{\frac{1 + I_{\text{ext}}(s)}{I_a(s)}} \quad (\text{II.2})$$

The refined background was obtained as

$$I_{\text{ext}}(s) = \left[\frac{M_t(s) - 1}{R \cdot M_{\text{exp}}(s)} \right] \cdot \frac{1}{I_a(s)} \quad (\text{II.3})$$

I_{ext} was fit to a polynomial to ensure a smooth function.

A correlation background B_{corr} can be defined which is the background which would bring the experimental curve into perfect agreement with the theoretical:

$$M_t(s) = R \left[\frac{I_x(s)}{B_{\text{corr}}(s)} - 1 \right] \quad (\text{II.4})$$

$$B_{\text{corr}}(s) = \frac{I_x(s)}{[(M_t(s)/R) + 1]} \quad (\text{II.5})$$

$$\text{Also, } B_{\text{corr}}(s) = \phi(s)[I_a(s) + I_{\text{ext}}(s)] \quad (\text{II.6})$$

$$\therefore \phi(s) = \frac{B_{\text{corr}}(s)}{I_a(s) + I_{\text{ext}}(s)} \quad (\text{II.7})$$

where I_{ext} is the smooth curve found in (II.3)

The sector function was determined for several CS_2 plates taken

at each distance. An average was determined for each distance to account for random errors in I_x . The individual determinations agree with the average to better than 2%. The long and short distance sector functions are given in Appendix A.

Digitized Data Collection

The diffraction patterns were recorded on Kodak electron image plates (15) of size 10.2 x 12.7 cm. A double beam Jarrell-Ash microdensitometer modified with a rotating stage and interfaced to a digital recorder (16) was used to obtain transmittances at 100 micron intervals across the pattern for the short camera position, and 150 microns for the long camera. The center of the diffraction pattern was located to within 0.001 mm by minimising the sum of the squares of the deviations between equivalent readings in each half of the scan. A correction for backlash in the drive screw was made to the displacements associated with the transmittance data. Typical values of the drift per reading are of the order of 0.0002 mm. Optical densities were calculated from the transmittance T by

$$O.D. = -\log T \quad (II.8)$$

Chapter III

DATA ANALYSIS

Scale Calibration

Initial calculations for the scale parameters L, λ were based on the scattering patterns due to solid MgO . The diameters of the sharp rings were measured with a travelling microscope, each ring being independently measured across at least four different diameters.

Magnesium oxide forms a face centered cubic lattice with lattice constant $a_0 = 4.2117 \pm 0.0005 \text{ \AA}$ (17), and a temperature dependence

$$a_0 = a_0 (0.999695 + 0.0000122T)$$

where T is in degrees absolute. The initial value for L was measured using a cathetometer. Each ring of radius r was assigned a set of Miller indices (h,k,l) and the trial values for L, λ were varied to bring the experimental scattering variable s into agreement with the theoretical value:

$$s(\text{experimental}) = (4\pi/\lambda) \sin(0.5 \tan^{-1}(r/L)) \quad (\text{III.1})$$

$$s(\text{theoretical}) = 4\pi(h^2 + k^2 + l^2)^{1/2}/2a_0 \quad (\text{III.2})$$

Typical calibration data for the long distance position are given in Appendix B. Final values for the scale parameters were determined by comparing the experimental scattering from CS_2 with that calculated from

the known structure (14). Optical densities from the two halves of the microdensitometer scan were then interpolated at integral values of the scattering variable q :

$$q = 40/\lambda \sin (0.5 \tan^{-1} (r/L)) = \frac{10}{\pi} S \quad (\text{III.3})$$

where r is the distance of a point on the plate to the center of the pattern.

The relativistic corrected wavelength is related to the voltage V by

$$\lambda = [150/V (1 + 9.834 \times 10^{-7})]^{1/2} \quad (\text{III.4})$$

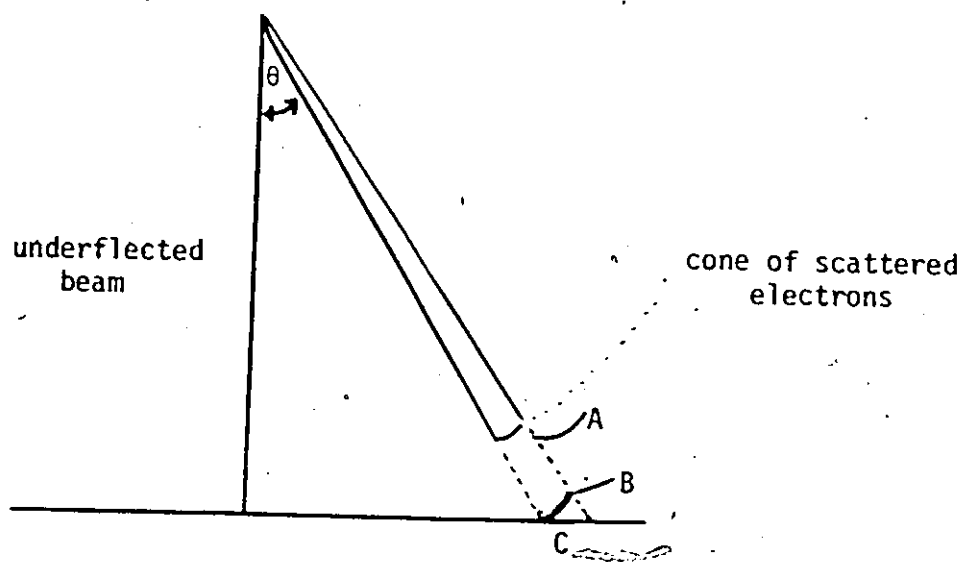
The mean voltage calculated from 13 calibration experiments taken over a period of three months was 57.88 kV with a standard deviation during that time of 0.13 kV. The MgO patterns showed no evidence of elliptical diffraction rings that might be caused by extraneous magnetic fields. The diffraction laboratory is situated in a ferro-concrete building and the effect of terrestrial magnetism is expected to be negligible (18).

The Experimental Molecular Scattering Curve

A linear density-intensity relationship was assumed for densities less than about 0.8 (19) and a correction was applied for darker plates (20). The electrons are scattered in a cone but are recorded on a flat photographic plate. To correct the intensities for this effect the following expression was used:

$$I(q) = I_0(q)/\cos^3\theta \quad (\text{III.5})$$

where θ is the angle between the incident and scattered wave vectors, I_0 is the intensity of electrons at the cone surface, and I is the intensity of electrons at the photographic plate. The relationship is shown in the following diagram.



B is related to A as $B = A/\cos^2\theta$

$C = B/\cos\theta = A/\cos^3\theta$

$$A_i = \frac{\sum_q (\text{Unscaled Average})}{\sum_q (\text{Data set } i)} \quad (\text{III.6})$$

The intensities of each data set were then multiplied by the corresponding scale factor and the resultant sets were numerically averaged to obtain a composite curve.

A smooth background was drawn through the oscillations in each composite curve to obtain the experimental molecular scattering component. Data from the two camera positions were spliced together (20) to form one continuous experimental curve from which the radial distribution function was calculated. Errors in the non-structural regions are due to a poorly drawn background and were useful for obtaining corrections to the trial background (21). The difference between the theoretical and experimental radial distribution function was then reinverted to obtain the correction to the molecular scattering:

$$(sM_{\text{expt}}(s)) = \int_0^{r_{\text{max}}} (f_{\text{theor}}(r) - f_{\text{exp}}(r)) \sin(sr) dr \quad (\text{III.7})$$

The corrected scattering curve was then used to determine the refined background.

Least Squares Analysis

Initial structure refinements were made on the radial distribution curve because it can be readily interpreted in terms of molecular parameters, and also because of an economy in computing costs. However, since it is necessary to splice in theoretical data from $s = 0$ to s_{\min} , the experimental radial distribution is to some extent dependent on the assumed structure. Final refinements were based on the experimental molecular scattering curve.

The least squares method was used in the manner proposed by Hedberg and Iwasaki (22). An algorithm was used which allows dependent internuclear distances to be calculated from a set of independent geometric parameters (23), usually bond lengths and angles. The transformation may be represented as

$$RP = R \quad (\text{III.8})$$

where P is the vector of independent parameters and R is the $N \times N$ matrix of internuclear distances. In order to calculate elements of the Jacobian matrix A , a matrix D is defined for the k th parameter:

$$D_{ij}^k = (\partial R_{ij} / \partial P_k) \quad (\text{III.9})$$

The derivative of the intensity curve with respect to the inter-nuclear distance is

$$H_{ij} = (\partial S M(s) / \partial R_{ij}) \quad (\text{III.10})$$

The elements of H are obtained by differentiation:

$$H_{ij} = \frac{2sf_i f_j \cos(\eta_i - \eta_j) \exp(-1/2 l_{ij}^2 s^2)}{\sum_i (f_i^2 + S_i)} \left[\frac{\cos(X)}{sr_{ij}} \left\{ 1 + \left(\frac{l_{ij}}{r_{ij}} \right)^2 \right\} \frac{\sin(X)}{s^2 r_{ij}^2} \right] \quad (\text{III.11})$$

$$\text{where } X = sr_{ij} - l_{ij}^2 - a_{ij} l_{ij}^4 s^3 / 6$$

The derivative of the intensity curve with respect to the mean square amplitudes is given by

$$H_{ij} = \frac{-2sf_i f_j \cos(\eta_i - \eta_j) \exp(-1/2 l_{ij}^2 s^2)}{\sum_i (f_i^2 + S_i) sr_{ij}} \left[\sin(X) s l_{ij} - \cos(X) \left(\frac{2l_{ij}}{r_{ij}} + \frac{a_{ij} l_{ij}^3 s^2}{2} \right) \right] \quad (\text{III.12})$$

The matrix D is then defined using

$$D = (T[P + \Delta P] - TP) / \Delta P \quad (\text{III.13})$$

The Jacobian may then be defined as

$$A = \text{Trace (HD)} \quad (\text{III.14})$$

The Weight Matrices

When only two sets of data were available a unit diagonal matrix with exponential weighting at the beginning and end of the data was used:

Weight	Data Region	
$\exp(-a_1[s-s_1]^2)$	$s_{\min} < s < s_1$	
1	$s_1 < s < s_2$	
$\exp(-a_2[s_1-s_2]^2)$	$s_2 < s < s_{\max}$	(III.15)

When three or more sets of data were available the diagonal weight matrix was composed from estimates of the variances of the individual observations. The sample variance V is given by

$$V = \frac{1}{n-1} \sum_i (x_i - \bar{x})^2 \quad (\text{III.16})$$

where n is the number of data sets

x_i is a scaled data set

\bar{x} is the average curve.

An estimate for the best weight of each data point can be made from $1/V$.

Correlations between the uncertainties of adjacent data points were taken into account by an off-diagonal weighting calculated in the manner proposed by Murata and Morino (24).

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PART II

THE MOLECULAR STRUCTURES OF $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$,
 CH_3GeF_3 AND CH_3GeBr_3

PART II
CHAPTER I

INTRODUCTION

The induction of double-bond character by the partial ionic nature of another bond has been used by Pauling (1) to explain the observed shortening of the carbon-halogen bond with progressive substitution of fluorine atoms. Resonance hybrids are proposed as the major contribution. Similar effects have been observed in silicon compounds and resonance hybrids have been proposed (2) using d-orbitals not available in the case of carbon. Ebsworth (3) gives a review of information regarding d-orbital participation in π bonding which has been extensively considered for second, but not third row elements.

The induced formation of partial double-bond character in the other group IV elements would be expected to occur with increasing difficulty, and there have been different interpretations drawn from a variety of techniques. The photoelectron spectra of the series $(\text{MH}_3)_2\text{Y}$ and MH_3SH ($\text{M} = \text{S}, \text{Si}, \text{Ge}$; $\text{Y} = \text{O}, \text{S}, \text{Se}, \text{Te}$) have been interpreted (4) on the basis of $(\text{p} - \text{d})\pi$ bonding with $\text{M} = \text{Si}$ and Ge . Evidence for $(\text{p} - \text{d})\pi$ bonding between chlorine and silicon or germanium has also been drawn from the photoelectron spectra of the silyl and germlyl halides (5). However, an ab initio molecular orbital calculation of the electronic structures of SiH_3F and GeH_3F (6) indicates that the spectra could be attributed to changes in the molecular structure on ionisation. Furthermore, the electric dipole moments of the series GeH_3X ($\text{X} = \text{Cl}, \text{Br}, \text{I}$) follow expected trends and do not support the postulation of $(\text{p} - \text{d})\pi$ bonding (7).

At present there is not a great deal of structural information known about germanium compounds, and it seems worthwhile to investigate the effect of halogen substitution in some simple molecules. A study of the germanium-carbon bond was undertaken for $(\text{CH}_3)_2\text{GeH}_2$ and $(\text{CH}_3)_4\text{Ge}$ by the vapor phase electron diffraction method to confirm the microwave structure of dimethylgermane (8) and to complete the methylgermane series. The molecular structures of CH_3GeF_3 and CH_3GeBr_3 have been examined to determine the effect of halogen substitution on the Ge-C and Ge-X bonds.

CHAPTER II

EXPERIMENTAL

The sample of tetramethylgermane was purchased from Alfa Inorganics Inc., while samples of $(\text{CH}_3)_2\text{GeH}_2$, CH_3GeF_3 and CH_3GeBr_3 were prepared and purified by Dr. R.T. Hemmings, to whom the author expresses his sincere thanks. Dimethylgermane was prepared (9) from LiAlH_4 reduction of dichloro(dimethyl)germane in $n\text{-Bu}_2\text{O}$. The purity was established from the vapor pressure (9), infra red and Raman bands (10) and from the reported ^1H n.m.r. spectrum (11). The purity of $(\text{CH}_3)_4\text{Ge}$ was verified by comparison with the reported vapor pressure (12), vibrational spectra (13) and ^1H n.m.r. spectrum (11). The sample of CH_3GeF_3 was prepared (14) from Pb(II)F_2 exchange with CH_3GeBr_3 and the purity established from the ^1H n.m.r. spectrum, infra red and Raman bands, sublimation pressure and melting point (15). The tribromide was prepared (16) by HBr exchange with CH_3GeCl_3 at room temperature and its purity was checked by ^1H n.m.r. (17), and infra red and Raman spectroscopy (18). The impurities in each sample were estimated to be less than 2%.

Sectored electron diffraction photographs were recorded on 4 x 5 in. Kodak Electron Image Plates using the instrument described in Part I. An accelerating voltage of about 58kV for nozzle to plate distances of 296.16 mm and 95.45 mm as measured with a cathetometer provided useful data covering the range $7 < q < 130 \text{ \AA}^{-1}$ where $q = (40/\lambda)\sin(\theta/2)$. Scale factors were obtained from CS_2 diffraction patterns taken concurrently with the gas patterns. The temperature at the nozzle was about 20°C . Tracings of the diffraction patterns were obtained using a Jarrell Ash microphotometer modified with a rotating stage and digital output (19). The relative intensities interpolated at integral q for $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3 are reproduced in Appendices C - F respectively.

CHAPTER III

ANALYSIS

The relative intensities obtained from traces of different plates taken under the same conditions were scaled and averaged to reduce the effect of random fluctuations. The experimental molecular intensity was calculated from each composite curve according to

$$M_{\text{exp}}(s) = \frac{s^4}{I_a(s)} \cdot \frac{I_x(s)}{\phi(s)} - B(s)$$

where $I_x(s)$ is the experimental sector intensity

$I_a(s)$ is the theoretical atomic scattering

$B(s)$ is the experimental background

$$s = \pi q/10$$

The curves were then spliced together (20). The experimental molecular intensity curve was fitted by a least squares procedure (19) which incorporated the Hartley modification (21). A diagonal weight matrix was based on the variance of each data point, and correlations of errors between data points was allowed for by means of a non-diagonal weight matrix (22).

The scattering factors used were obtained by a four-point interpolation to 58 kV from the values tabulated by Bonham, Schafer and Yates (23). Anharmonicity corrections were made for bonded atom

pairs (24,25). The structures were parametrized by means of a set of standard coordinate types (26). The calculated parameter values are based on r_g distances.

CHAPTER IV

RESULTS

In view of the microwave structure for dimethylgermane (8) and the tetrahedral geometry of tetramethylgermane, the choice for initial parameters in these molecules was routine. For the methyl trihalides, the input value for the Ge-C distance was that found in $(\text{CH}_3)_3\text{GeCl}$ (27), the Ge-F bond distance was taken from GeF_4 (28) and the Ge-Br bond distance from GeBr_4 (29). The final estimates for the bond distances and angles are presented in Table I. Estimates for the root mean square amplitudes of vibration are given in Table II. The errors are three times those calculated from the least squares analysis and are estimated to be at the 95% confidence level. Drawings of the structures of $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3 are presented in Figures 1 through 4 respectively.

The molecular scattering curve for dimethylgermane is reproduced in Figure 5. The radial distribution function which is shown in Figure 6 illustrates some of the problems encountered in the parameter refinements. The Ge-C bond distance is well resolved, but the contribution to the scattering curve from non bonded Ge...H distances is greater than that due to the bonded pairs C-H and Ge-H. Least squares refinements tend to fit the Ge...H contribution rather than the C-H and Ge-H parameters directly. When the CGeC angle was allowed to vary a value of 106 degrees was found, all other parameters remaining

Table I

The structural parameters^a and standard deviations for $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3

Parameter	$(\text{CH}_3)_2\text{GeH}_2$	$(\text{CH}_3)_4\text{Ge}$	CH_3GeF_3	CH_3GeBr_3
Ge-C	1.953 ± 0.003	1.955 ± 0.005	1.921 ± 0.012	1.87 ± 0.04
Ge-X	1.56 ± 0.04	-	1.708 ± 0.003	2.284 ± 0.002
C-H	1.074 ± 0.016	1.08 ± 0.03	1.14 ± 0.08	1.08^b
<CGeC	110.0^b	109.47^b	-	-
<CGeX	109.5^b	-	114.0 ± 0.1	111.2 ± 0.3
<XGeX	108.9^b	-	104.6 ± 0.9	107.7 ± 0.3

^aBond lengths are r_g values in Å units, angles in degrees

^bParameter constrained during least squares.

Table II

Root mean square amplitudes of vibration¹ and standard deviations for $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3

Parameter	$(\text{CH}_3)_2\text{GeH}_2$	$(\text{CH}_3)_4\text{Ge}$	CH_3GeF_3	CH_3GeBr_3
Ge-C	0.046 ± 0.004	0.044 ± 0.007	0.045^2	0.042 ± 0.06
Ge-X	0.08^2	-	0.051 ± 0.003	0.047 ± 0.002
C-H	0.078^2	0.078^2	0.078^2	0.078^2
Ge...H	0.079 ± 0.015	0.085 ± 0.029	0.09^2	0.1^2
C...C	0.09^2	0.086 ± 0.029	-	-
C...X	0.12^2	-	0.089 ± 0.021	0.076 ± 0.036
X...X	0.13^2	-	0.094 ± 0.016	0.105 ± 0.006

¹Parameters are in Å units

²Constrained during final least squares

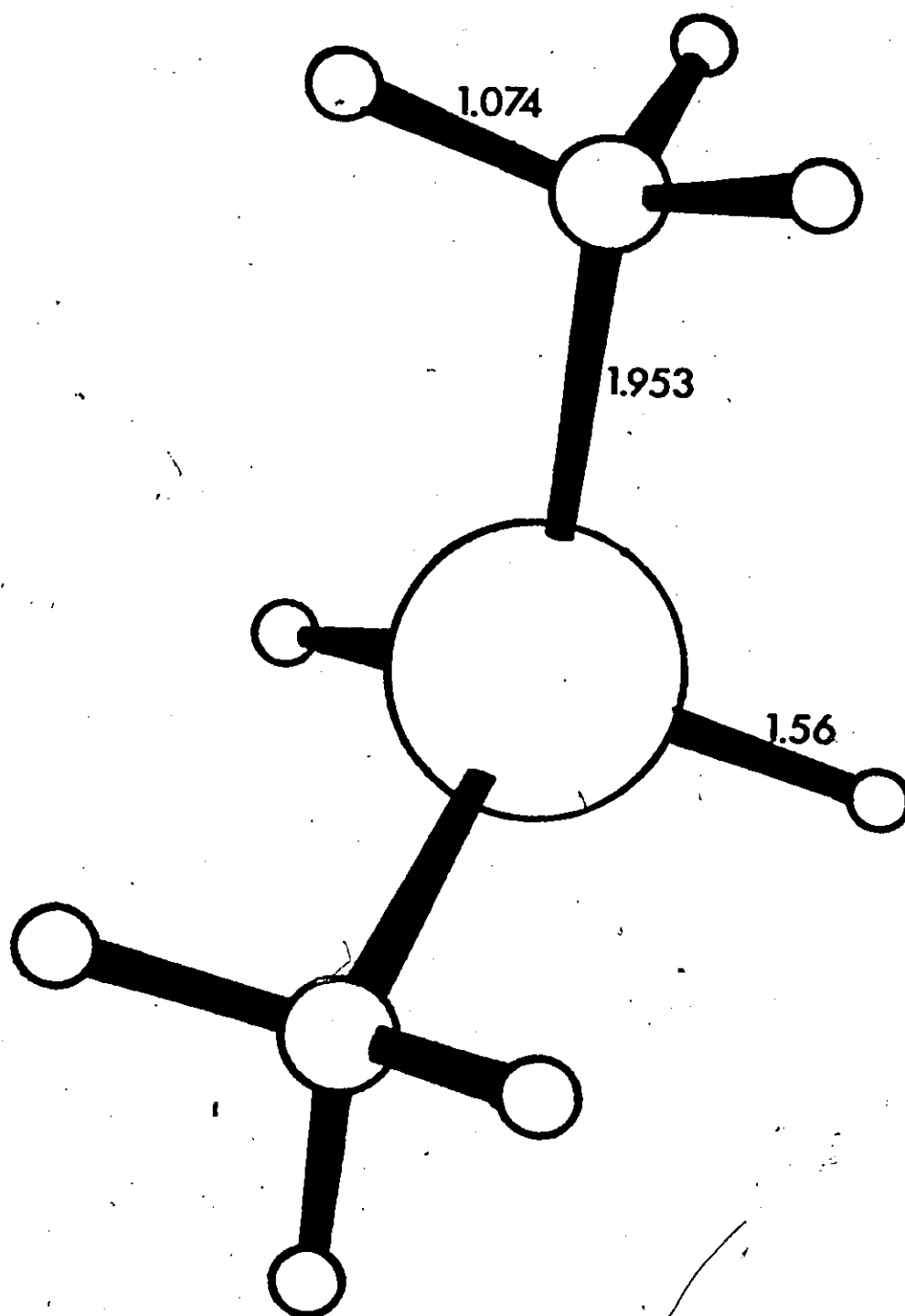


Figure 1. The Molecular Structure of Dimethylgermane

Figure 2

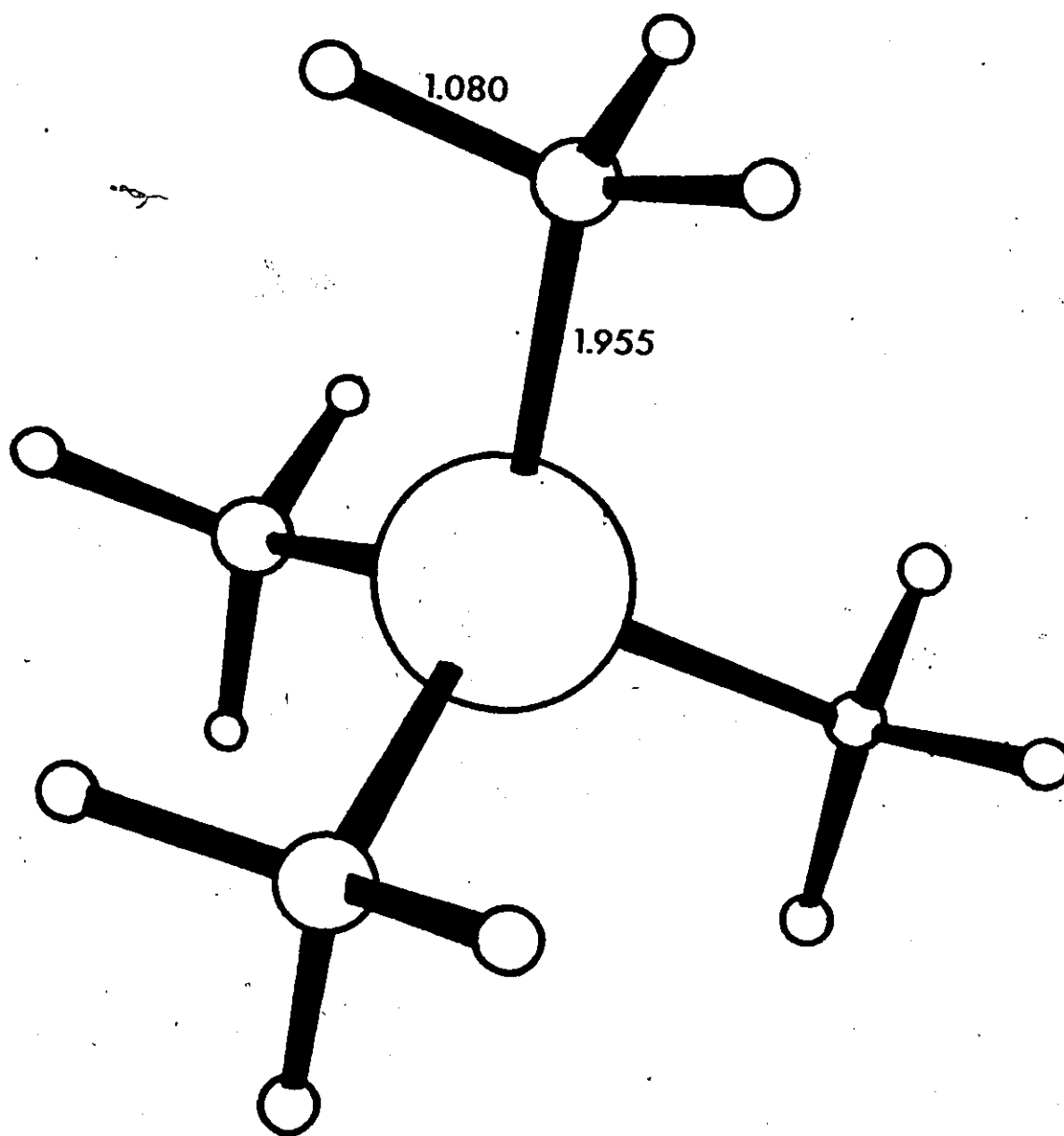


Figure 2. The Molecular Structure of Tetramethylgermane

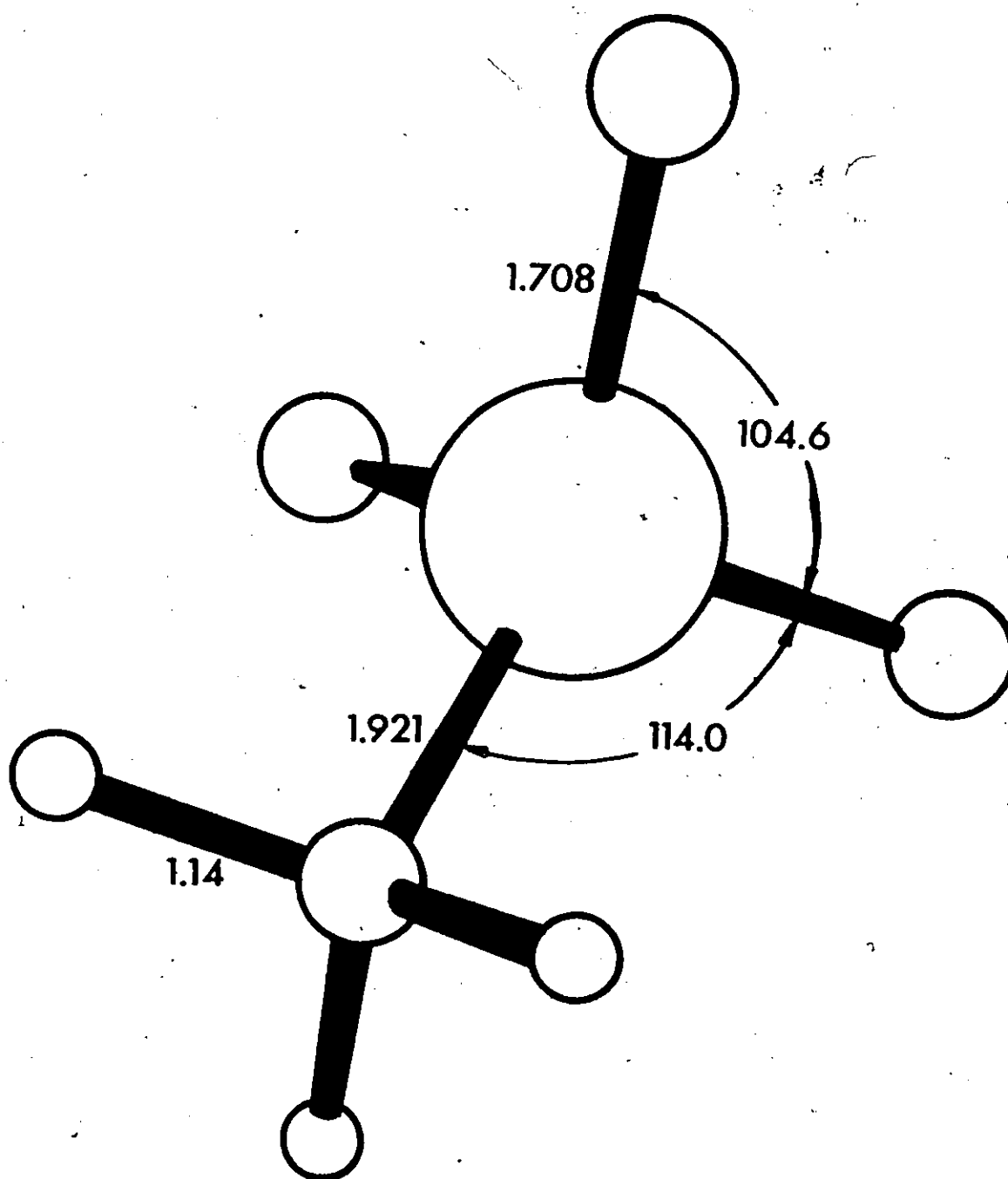


Figure 3. The Molecular Structure of Trifluoro(methyl)germane

Figure 4

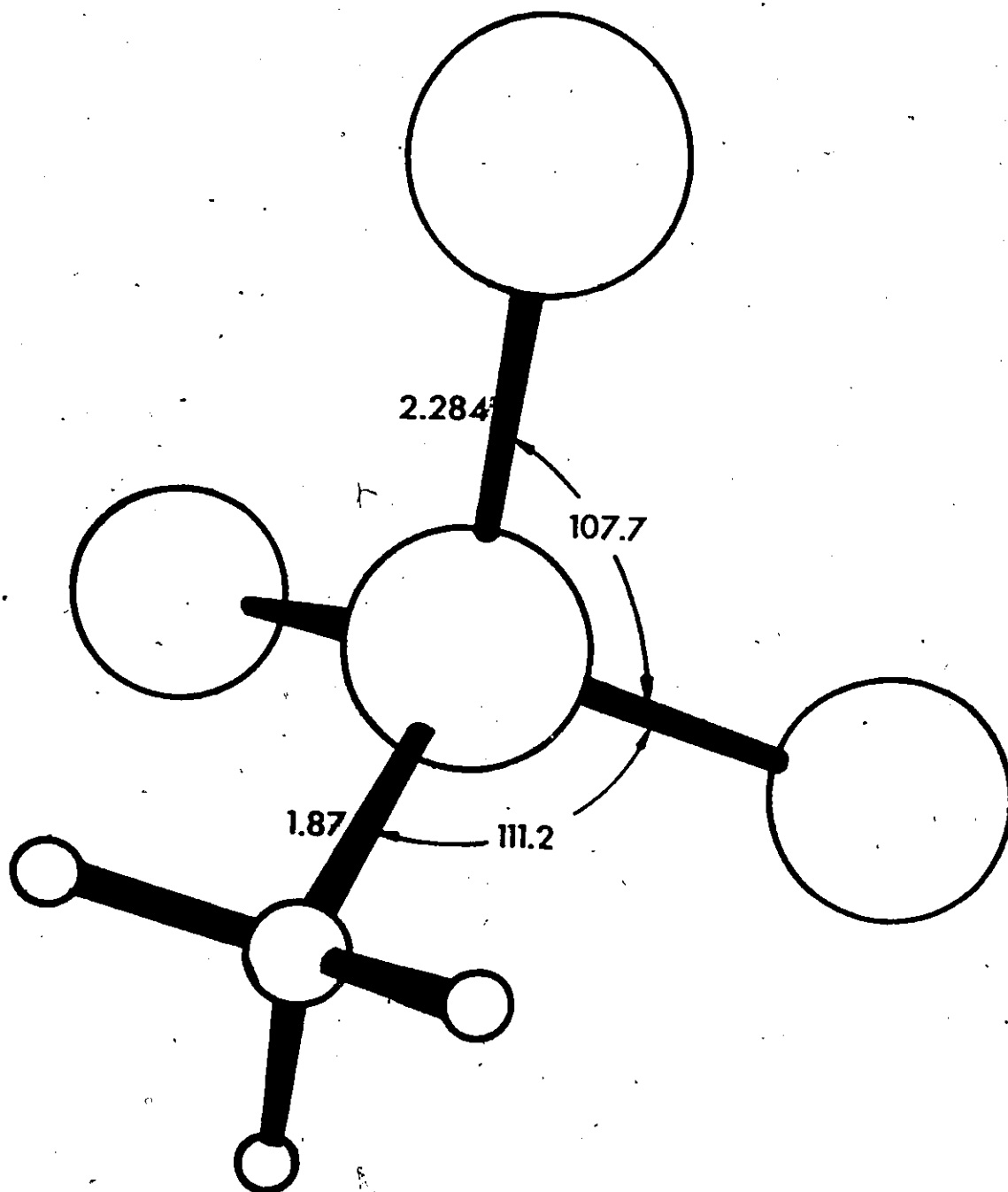


Figure 4. The Molecular Structure of Tribromo(methyl)germane

Figure 5

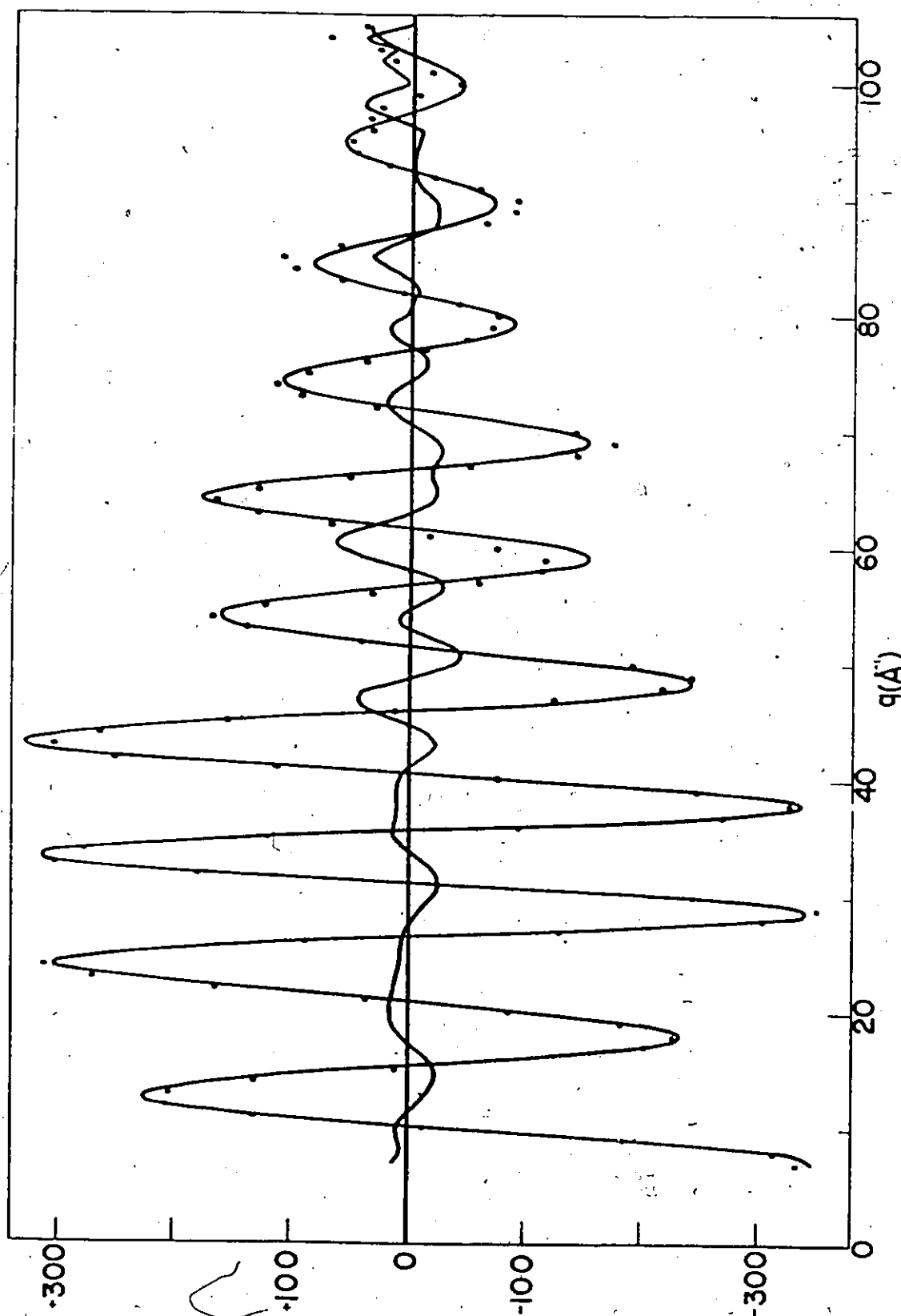


Figure 5. The Theoretical, Experimental (dots) and Difference Molecular Scattering Curve for Dimethylgermane.

Figure 6

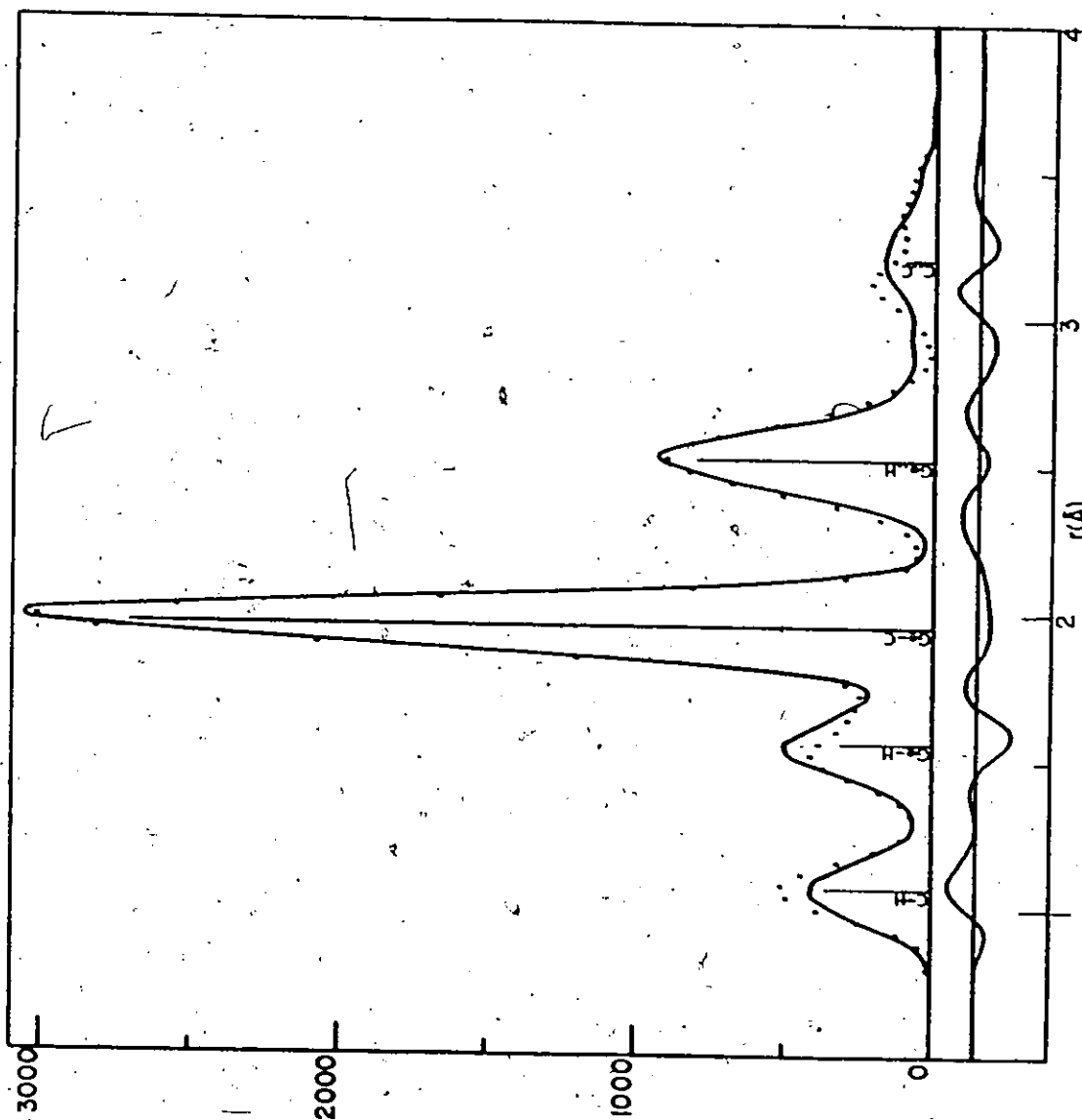


Figure 6. The Theoretical, Experimental (dots) and difference Radial Distribution Function for Dimethylgermane.

unchanged within the estimated errors. Accordingly, this angle was constrained to 110.0 degrees (8). There were no large correlations between the parameters.

The molecular scattering curve and radial distribution function for tetramethylgermane are shown in Figures 7 and 8 respectively. Once again, the radial distribution function illustrates that the non-bonded contributions of Ge..H and C..C are larger contributors to the molecular scattering than the C-H bond. Correlations between parameters presented no problems during parameter refinements.

The molecular scattering curve and radial distribution function for CH_3GeF_3 are reproduced in Figures 9 and 10 respectively. Although correlations between the parameters were small, the contribution due to the Ge-C bond at 1.92Å appears as a shoulder on the Ge-F bond. Attempts to simultaneously refine the Ge-C mean square amplitude resulted in an unrealistically large value of 0.068Å, all other parameters remaining at those given in Tables I and II, within the quoted uncertainties. The Ge-C mean square amplitude was therefore constrained to 0.045Å in accordance with the value for this parameter in di- and tetramethyl germane.

The molecular scattering curve and radial distribution function for CH_3GeBr_3 are illustrated in Figures 11 and 12 respectively. The radial distribution function shows that the Ge-C contribution is relatively small and that the non-bonded C..Br distance appears as a shoulder on Br..Br. Although correlations between the parameters

Figure 7

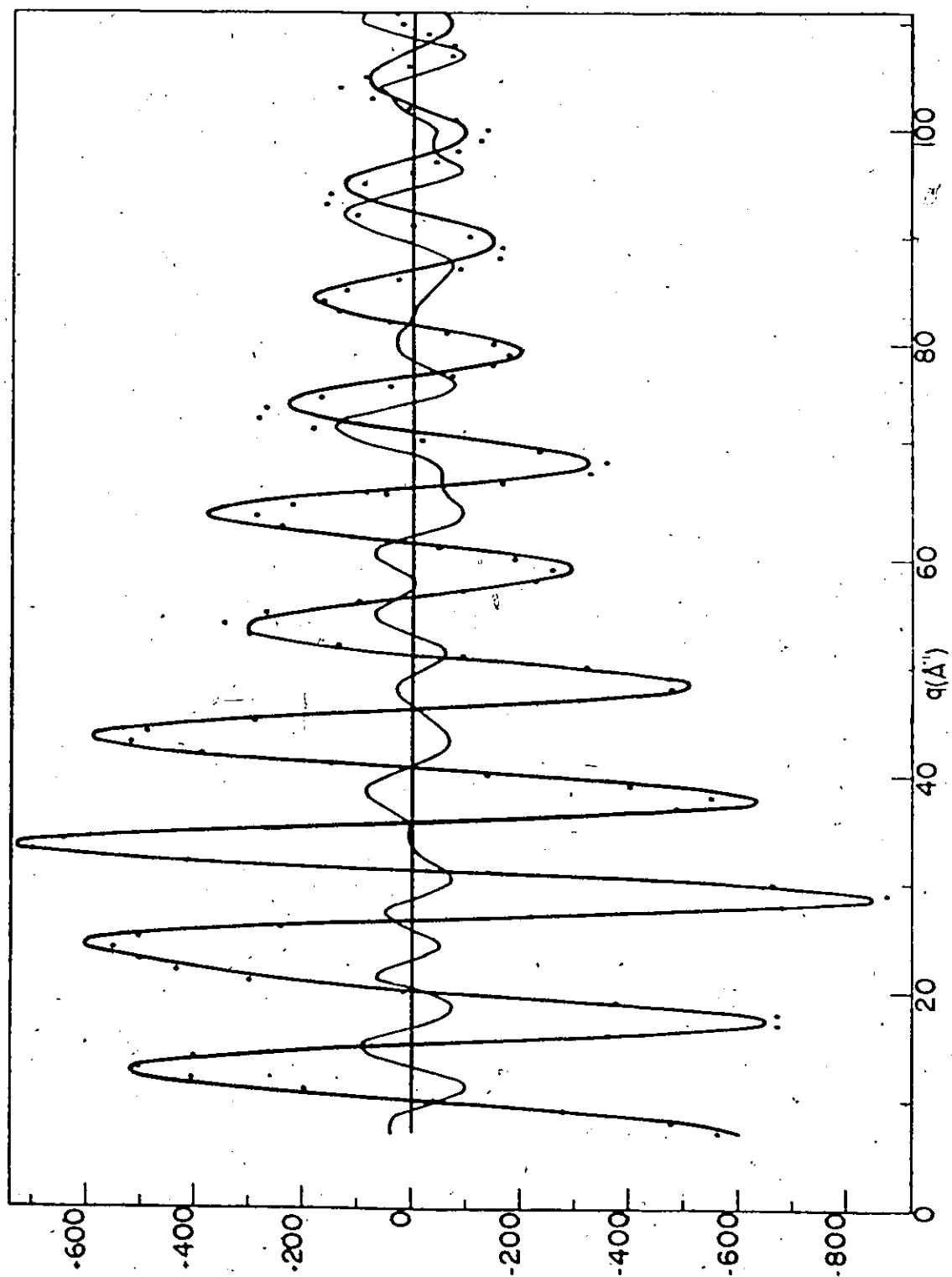


Figure 7. The Theoretical, Experimental (dots) and Difference Molecular Scattering Curve for Tetramethylgermane.

Figure 8

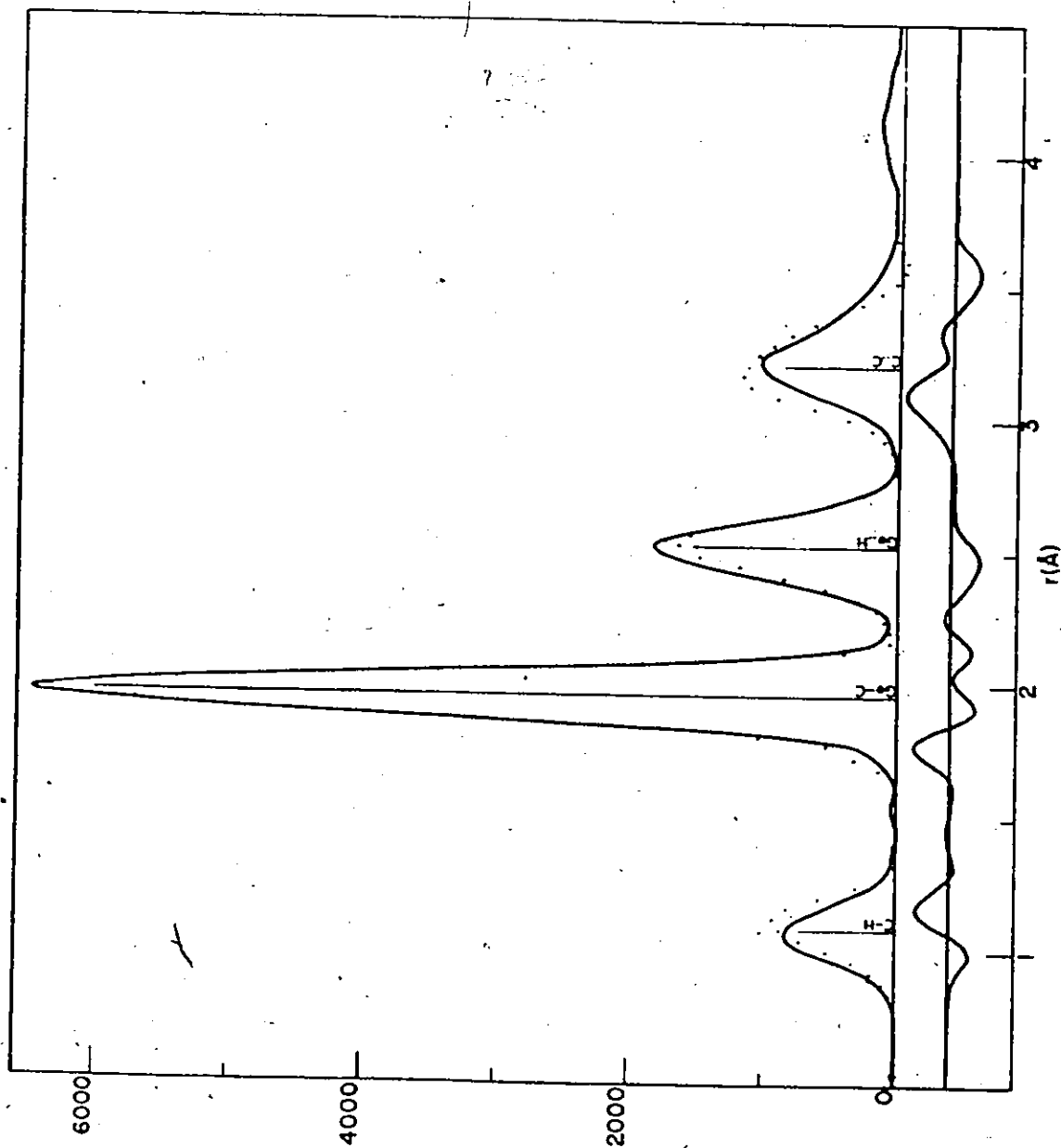


Figure 8. The Theoretical, Experimental (dots) and Difference Radial Distribution Function for Tetramethylgermane.

Figure 9

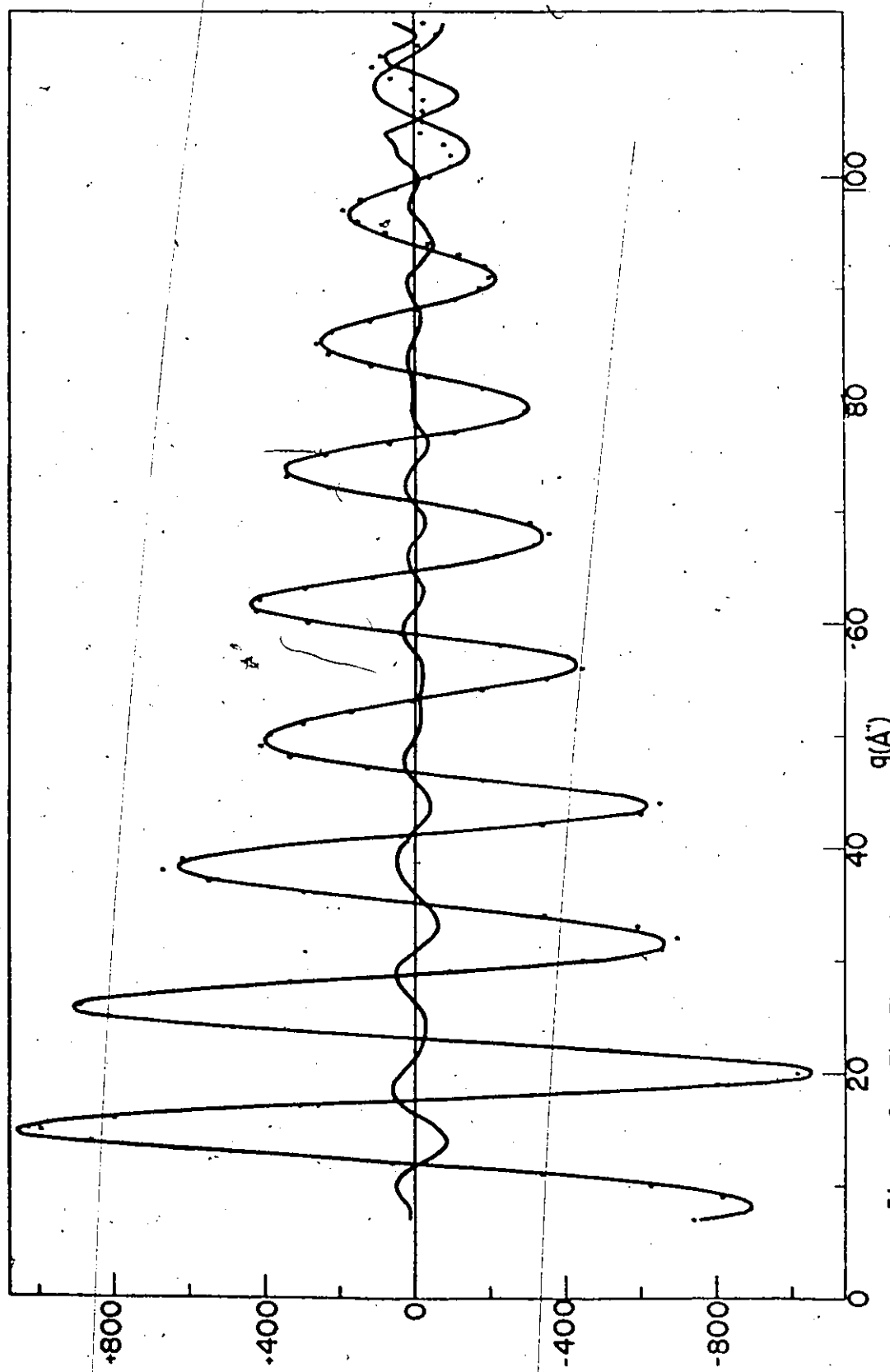


Figure 9. The Theoretical, Experimental (dots) and Difference Molecular Scattering Curve for Trifluoro(methyl)germane

Figure 10

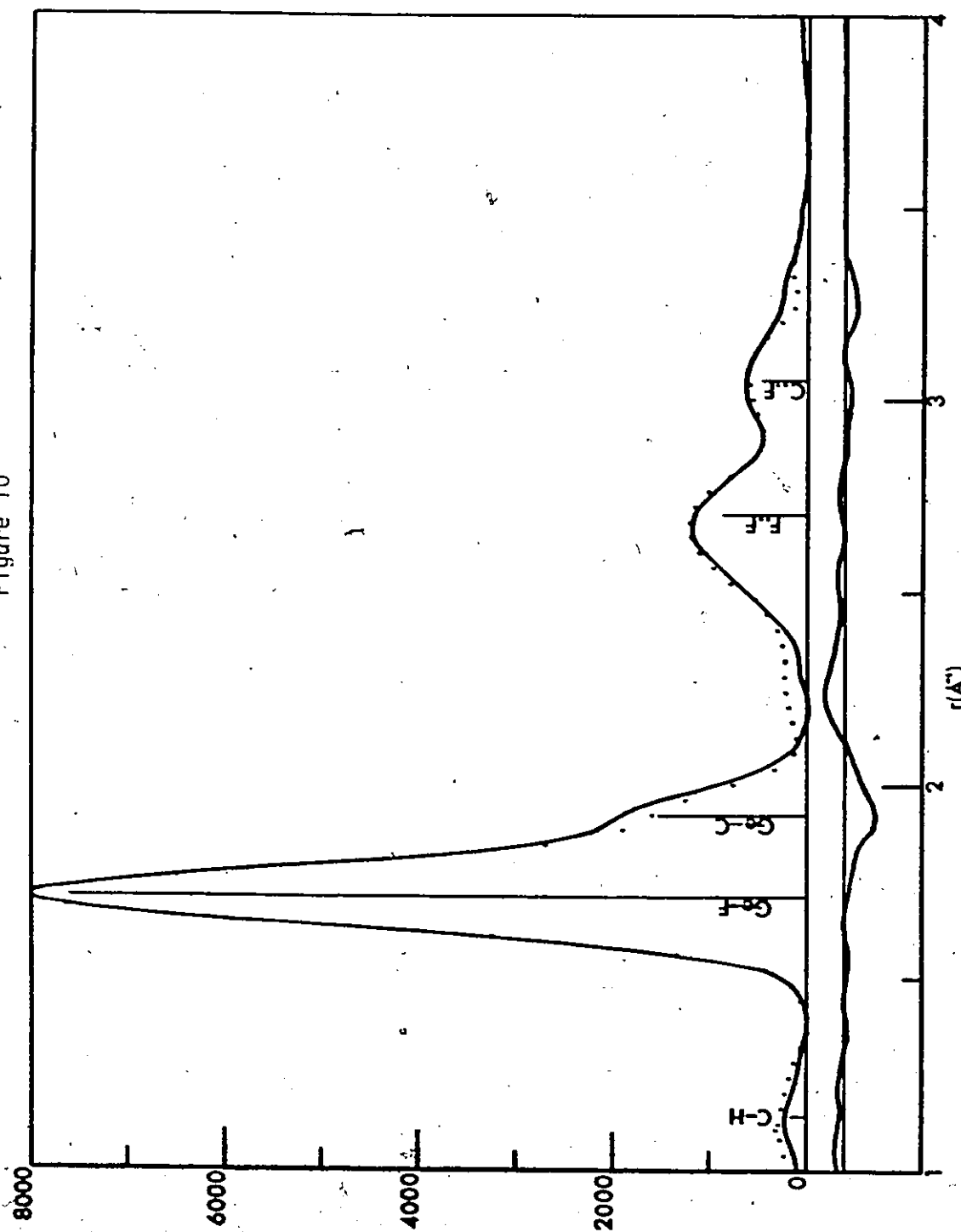


Figure 10. The Theoretical, Experimental (dots) and Difference Radial Distribution Function for Trifluoro(methyl)germane.

Figure 11

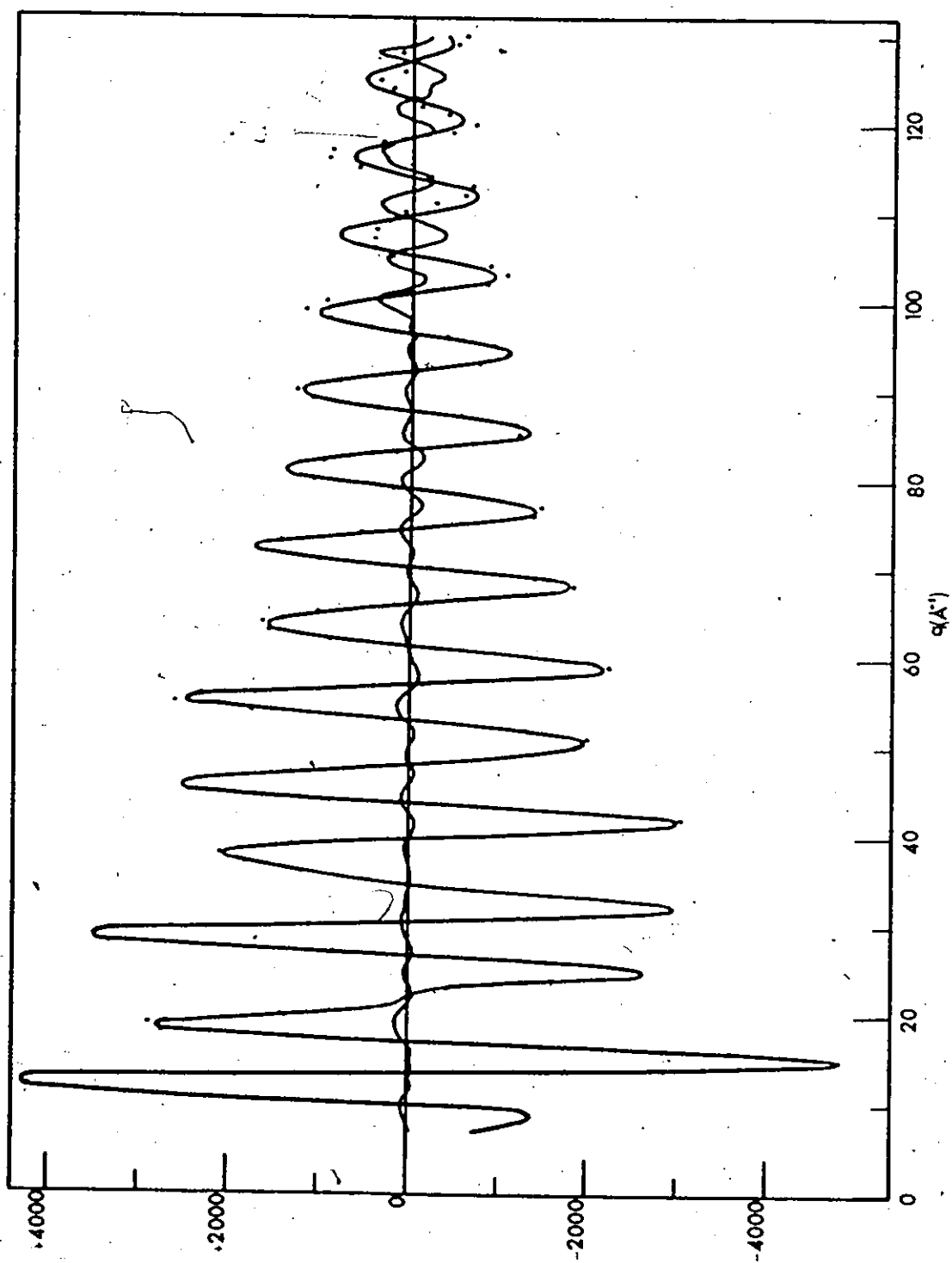


Figure 11. The Theoretical, Experimental (dots) and Difference Molecular Scattering Curve for Tribromo(methyl)germane.

Figure 12

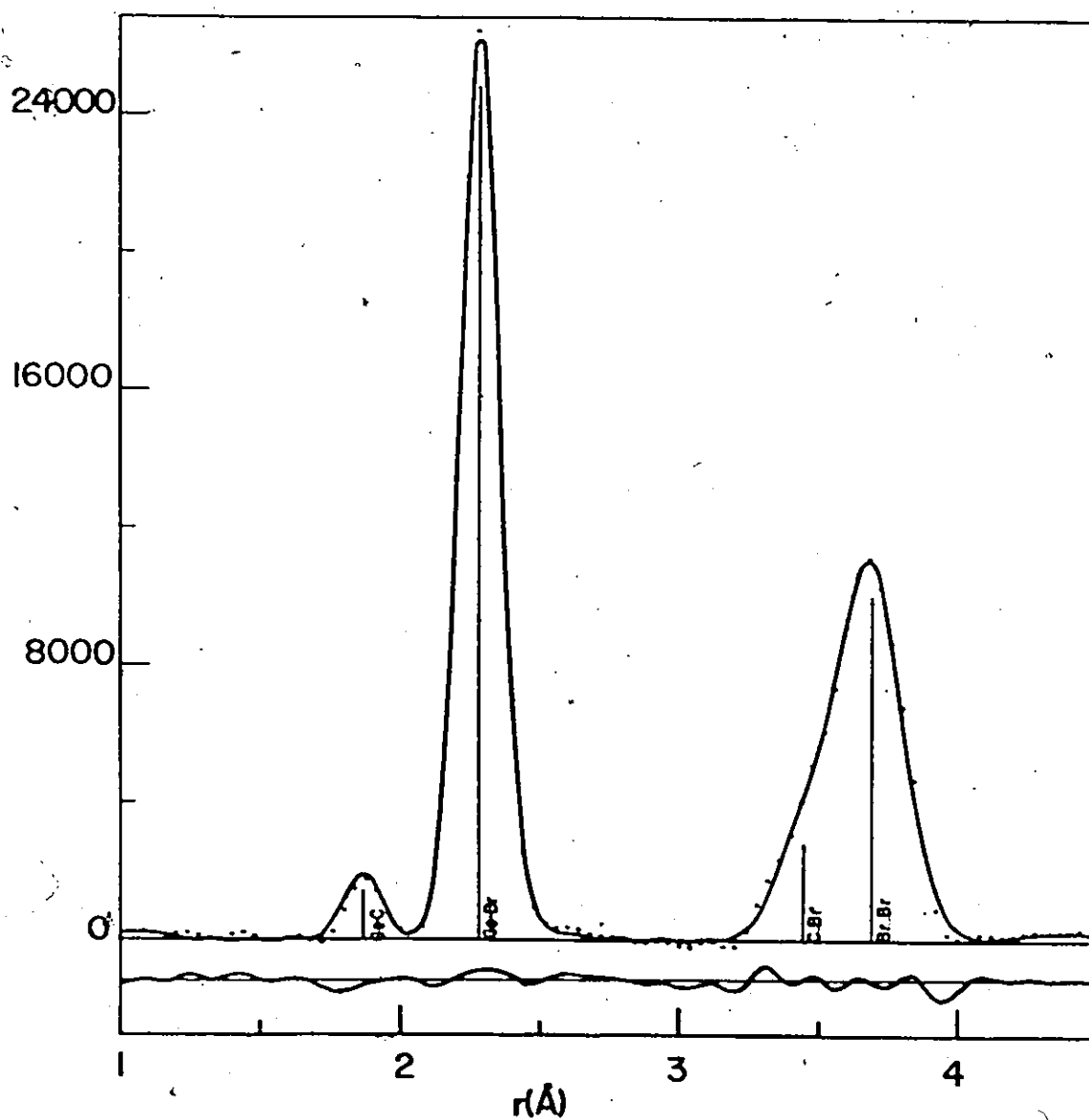


Figure 12: The Theoretical, Experimental (dots) and Difference Radial Distribution Function for Tribromo(methyl)germane

were small, the Ge-C bond cannot be accurately determined. The contribution due to C-H is of the order of the error curve in this molecule, and the parameter was constrained during least squares refinements.

The non-bonded interatomic distances (excluding those involving hydrogen atoms bonded to carbon) are given in Table III, and the correlation matrices derived from the least squares refinement are listed in Appendix G.

Table III.

Non-bonded Interatomic Distances of the Heavy Atoms¹ in $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3

	$(\text{CH}_3)_2\text{GeH}_2$	$(\text{CH}_3)_4\text{Ge}$	CH_3GeF_3	CH_3GeBr_3
X...X	2.628	-	2.703	3.688
C...X	2.851	-	3.046	3.435
C...C	3.200	3.193	-	-

¹Hydrogen atoms bonded to carbon are excluded

CHAPTER V

Discussion on the Metal-Carbon Bond for the Group IV Elements

Listed in Table IV are the available M-C bond lengths in the series $M(\text{CH}_3)_{4-n}\text{H}_n$ where $M = \text{Si}, \text{Ge}, \text{Sn}, \text{Pb}$. Bond lengths (r_s) deduced from microwave studies of isotopically substituted molecules are close to the equilibrium values r_e (40). The thermally averaged distance r_g which can be determined from electron diffraction may be corrected to an approximate r_e value (24), which is comparable to within a few thousands of an angstrom to the r_s value:

$$r_e = r_g - \frac{3}{2} a_{ij} l_{ij}^2 \quad (1)$$

The anharmonic constant a_{ij} used for the Ge-C bond was 2\AA^{-1} ; l_{ij} is the estimated root mean square amplitude of vibration. Table IV shows excellent agreement of the M-C bond in the series $M(\text{CH}_3)_{4-n}\text{H}_n$ ($M = \text{Si}, \text{Ge}$) as determined by electron diffraction and microwave when to an approximate common bond length definition (r_e) is made. The Ge-C bond length in tetramethylgermane as determined from the present study represents a considerable improvement over an earlier value of $1.98 \pm 0.03\text{\AA}$ (41).

An unweighted estimate for the mean equilibrium Si-C bond length in the methyl silanes is 1.868\AA with an estimated standard deviation of 0.001\AA . The corresponding values for the Ge-C bond are 1.948 and 0.001\AA . It is readily apparent that the Ge-C bond in the methyl germanes exhibits a constant value as is the case for the Si-C bond in the methyl silanes.

Table IV

M-C Bond Lengths in $M(\text{CH}_3)_4$, $M = \text{Si, Ge, Sn, Pb}$

CH_3MH_3		$(\text{CH}_3)_2\text{MH}_2$		$(\text{CH}_3)_3\text{MH}$		$(\text{CH}_3)_4\text{M}$	
Si-C	$r_s = 1.867 \pm 0.001$ (30)	$r_s = 1.867 \pm 0.002$ (31)	$r_s = 1.868 \pm 0.002$ (32)	$r_e = 1.869 \pm 0.003$ (33)			
				$r_g = 1.878 \pm 0.002$			
Ge-C	$r_s = 1.945 \pm 0.001$ (34)	$r_s = 1.950 \pm 0.003$ (8)	$r_s = 1.947 \pm 0.006$ (35)	$r_e = 1.949 \pm 0.005$ (this work)			
		$r_e = 1.947 \pm 0.003$ (this work)		$r_g = 1.955 \pm 0.005$			
		$r_g = 1.953 \pm 0.003$ (this work)					
Sn-C	$r_s = 2.143 \pm 0.002$ (36)	2.150 ± 0.003 (37)	2.147 ± 0.004 (37)	$r_g = 2.134 \pm 0.007$ (38)			
Pb-C				$r_e = 2.229 \pm 0.009$ (39)			
				$r_g = 2.238 \pm 0.009$			

The r_g (Si-Si) bond length in hexamethyldisilane (33) is $2.342 \pm 0.009\text{\AA}$ ($\ell(\text{Si-Si}) = 0.068\text{\AA}$) and in disilane (42) it is $2.331 \pm 0.003\text{\AA}$ ($\ell(\text{Si-Si}) = 0.046\text{\AA}$). The r_e bond distances are thus 2.328 and 2.325 \AA respectively. An estimate for the bonding radius of silicon can then be made as $1.163 \pm 0.003\text{\AA}$. Based on the C-C bond length in ethane (43) the bonding radius of carbon is $0.767 \pm 0.001\text{\AA}$. Simple additivity of radii yields a predicted Si-C bond length of $1.930 \pm 0.004\text{\AA}$, a value 0.062 \AA larger than determined in the methyl silanes. This shortening and the possibility of it being due to $p\pi - d\pi$ interaction has been pointed out previously (33).

Similar arguments can be made for the Ge-C bond. The r_g (Ge-Ge) distance in digermane (44) is $2.404 \pm 0.003\text{\AA}$. Given the observed mean square amplitude of vibration of 0.047 ± 0.012 , the correction to an r_e (Ge-Ge) distance is thus 2.397 \AA , and the approximate bonding radius for germanium is then 1.199 \AA . Using this value, the predicted Ge-C bond length is 1.966 \AA which is 0.018 \AA larger than experimentally found. This is a much smaller difference than the 0.062 \AA found for the Si-C bond, and indicates that although the M-C bond length does not vary with the number of methyls substituted in either the germanes or silanes, the nature of the bonding in the two cases may be different.

Despite objections (45) that Pauling's electronegativity values (1) for silicon and germanium are based on bonds which may involve some (p-d) π bonding, the unbiased electronegativities of Si and Ge that have been calculated (45) are still approximately equal. It may therefore be instructive to use Pauling's scheme on a quantitative basis to compare

The Si-C and Ge-C bonds. The electronegativity differences, 0.7, is the same for both pairs and leads to a prediction of 12% ionic character in both bonds (1). The shortening calculated by the Schomaker-Stevenson relationship (46) is 0.042 Å for Si-C and 0.028 Å for Ge-C. This leads to a prediction for the Si-C bond length of 1.888 Å, which is larger by 0.020 Å than experimentally found. This difference is of the order of the reliability of the Schomaker-Stevenson equation, and cannot be relied on to suggest bond shortening by (p-d) π interaction in the Si-C bond. It is of note that ab initio LCAO-MO calculations (47) suggest that 10% of the total overlap population for the Si-C bond in methylsilane is due to (p-d) π interaction. The calculated value for the Ge-C bond length is 1.938 Å, which is 0.010 Å shorter than experimentally determined, but well within the reliability of the Schomaker-Stevenson relationship. No evidence for d-orbital involvement can be drawn from the Ge-C bond length.

The electron diffraction technique cannot accurately determine the C-Ge-C angle in $(\text{CH}_3)_2\text{GeH}_2$. However, the microwave study (8) has determined this angle to be slightly larger than tetrahedral, and a similar result was found for the angle C-Si-C in dimethylsilane (31). It is likely that the small difference in electronegativity of hydrogen as compared to carbon is not enough to affect the bond angles in the manner proposed by Bent (48) and that the major influence is due to non-bonded interactions (49). The Van der Waals radius of the methyl group is about 2.0 Å as compared to about 1.2 Å for hydrogen.

In the methylstannane series only the structure of CH_3SnH_3 has been published to date (36). The Sn-C bond length was found to be

$r_e = 2.143 \pm 0.002 \text{ \AA}$, and it is interesting to note that preliminary values for the Sn-C bond length in $(\text{CH}_3)_2\text{SnH}_2$ and $(\text{CH}_3)_3\text{SnH}$ (37) are 2.150 ± 0.003 and $2.147 \pm 0.004 \text{ \AA}$ respectively. These preliminary values are probably electron diffraction results and, if so, are in excellent agreement with the Sn-C bond length in CH_3SnH_3 determined by the microwave method. The preliminary Sn-C bond length in tetramethylstannane (38) is $r_g = 2.134 \pm 0.007 \text{ \AA}$, which is rather less than expected from the results of other methylstannanes. The preliminary value for the Sn-C bond length in tetramethylstannane therefore is difficult to reconcile in view of the results for the methyl-silanes and germanes. However, discussion on the methylstannane series is without a firm foundation until final publication of their structures.

CHAPTER VI

Discussion on the Molecular Structures of CH_3GeF_3 and CH_3GeBr_3

One of the general features of structural chemistry is a progressive decrease in M-X bond length on increasing the amount of electronegative substitution. In the case of first-row elements, Bent (48) has inferred that the decrease in bond length which is observed with progressive electronegative substitution is due to an increase in s character of the bond. Both Bent's interpretation and the VSEPR theory (50) lead to quantitative predictions for the bond angles in molecules with atoms of different electronegativity. Table V shows a progressive decrease of the Si-F bond in the fluorosilane series and Table VI illustrates that increasing substitution of fluorine in the methylfluorosilane series also causes a shortening of the Si-C bond. This effect has been attributed (3) to resonance structures using the 3d-orbitals of silicon.

There is less structural data available for germanium compounds, and no series as complete as those listed in Tables V and VI. However, the trend is for progressive shortening of the Ge-X and Ge-C bond lengths with increasing amounts of halogen substitution. For instance, the Ge-Cl bond length in GeCl_4 (55) is $r_e = 2.106 \pm 0.004 \text{ \AA}$ and in ClGeH_3 (56) the value is $r_s = 2.148 \pm 0.003 \text{ \AA}$, a difference of $0.042 \pm 0.007 \text{ \AA}$. Using a value of 0.99 \AA for the single bond radius of the chlorine atom, and making the Schomaker-Stevenson electronegativity correction (46), the predicted Ge-Cl single bond length is 2.141 \AA , in excellent agreement with the result for ClGeH_3 . The decrease of the Ge-Cl bond length in the tetra-

Table V

The Si-F bond lengths of fluorinated silanes

Molecule	Si-F	Reference
SiH_3F	$r_s = 1.593 \pm 0.002$	51
SiH_2F_2	$r_s = 1.577 \pm 0.001$	52
SiHF_3	$r_s = 1.562 \pm 0.005$	53
SiF_4	$r_e = 1.546 \pm 0.002$	54
	$r_g = 1.552 \pm 0.002$	

Table VI

The Si-F and Si-C bond lengths of methyl fluorosilanes

Molecule	Si-F	Si-C	Reference
CH_3SiH_3	-	$r_s = 1.867 \pm 0.001$	30
$\text{CH}_3\text{SiH}_2\text{F}$	$r_s = 1.600 \pm 0.005$	$r_s = 1.848 \pm 0.005$	57
CH_3SiHF_2	$r_s = 1.583 \pm 0.002$	$r_s = 1.833 \pm 0.002$	58
CH_3SiF_3	$r_s = 1.574 \pm 0.007$	$r_s = 1.812 \pm 0.014$	59

chloride is 0.032\AA and might be due to some (p-d) π interaction. The reported Si-Cl bond lengths in SiCl_4 (60) and in ClSiH_3 (61) are $r_e = 2.013 \pm 0.004$ and $r_s = 2.048 \pm 0.001\text{\AA}$ respectively. The difference is 0.035\AA which, within the estimated errors, is the same as that observed in the germanium molecules. The predicted Si-Cl bond length is 2.079\AA , which is 0.03\AA longer than found in ClSiH_3 and might be due to a small amount of (p-d) π overlap.

The most precise determination of the Ge-C bond length in a molecule with halogen substitution to date is $r_s = 1.9400 \pm 0.0001\text{\AA}$ in $(\text{CH}_3)_3\text{GeCl}$ (62). This is a small (0.007\AA) but significant decrease of the Ge-C bond length as compared with that found in $(\text{CH}_3)_3\text{GeH}$ (35). The Ge-Cl bond length was found to be $r_s = 2.170 \pm 0.001\text{\AA}$ which is 0.064\AA longer than in GeCl_4 . For comparison, the Si-C and Si-Cl bond lengths in $(\text{CH}_3)_3\text{SiCl}$ have been determined (63) as $r_s = 1.857 \pm 0.005$ and $2.022 \pm 0.05\text{\AA}$ respectively. The large uncertainty in the Si-Cl bond length makes meaningful comparisons difficult, but the value for the Si-C bond length is 0.011\AA shorter than in $(\text{CH}_3)_3\text{SiH}$ (32), a shortening similar to that in the corresponding germanium compounds. However, there is a difference in the angles reported, $\angle\text{CGeCl}$ being 106.2° in $(\text{CH}_3)_3\text{GeCl}$ and $\angle\text{CSiCl}$ being 110.5° in $(\text{CH}_3)_3\text{SiCl}$. It should be noted that in $(\text{CH}_3)_3\text{SiCl}$ the CSiCl angle is reported to be larger than the CSiC angle, a relative difference which would not be predicted by either the VSEPR theory (50) or Bent's rules (48).

The molecular structure of CH_3GeF_3 further illustrates the power of increasing halogen substitution to shorten the Ge-C and Ge-X bond lengths.

The following values for the Ge-F bond length have been determined:

$r_s = 1.751 \pm 0.005$ in $\text{CH}_3\text{GeH}_2\text{F}$ (64), $r_s = 1.735 \pm 0.01$ in GeH_3F (65), $r_g = 1.708 \pm 0.003\text{\AA}$ in CH_3GeF_3 , and $r_g = 1.67 \pm 0.03\text{\AA}$ in GeF_4 (28). The predicted value of the Ge-F bond length, making the electronegativity correction (46) is 1.743\AA , which is 0.043\AA larger than the equilibrium distance found in this work. Making allowances for the crude approach taken in using the Schomaker-Stevenson equation, there still appears to be some shortening of the Ge-F bond in this molecule that might be due to (p-d) π interaction. Dipole moment studies of the series $(\text{CH}_3)_n\text{GeF}_{4-n}$ (66) indicates that (p-d) π bonding between germanium and fluorine is very small even in CH_3GeF_3 . In contrast, the predicted Si-F bond length is 1.881\AA which is 0.131\AA longer than that determined in CH_3SiF_3 (59), and can only be accounted for by a significant amount of (p-d) π bonding. A comparison of the bond angles indicates that the Ge-F bond in CH_3GeF_3 might be slightly more polar than the Si-F bond in CH_3SiF_3 . The FGeC angle is $114.0 \pm 0.1^\circ$ and the FSiC angle is $112.3 \pm 1.1^\circ$.

The Ge-C bond distance is $r_e = 1.915 \pm 0.012\text{\AA}$ in CH_3GeF_3 as compared to $r_s = 1.925 \pm 0.002\text{\AA}$ in $\text{CH}_3\text{GeH}_2\text{F}$ (64) and $r_s = 1.945 \pm 0.001\text{\AA}$ in CH_3GeH_3 (36). Some shortening of the Ge-C bond seems to take place with increasing fluorine substitution though the effect is not as large as that found in the corresponding silicon molecules. Further indications of this effect have been observed from force constant studies on the methyl(fluoro)germanes (14). The Ge-C and Ge-F force constants progressively increase with increasing fluorine substitution. The Si-C bond distance in CH_3SiF_3 is $r_s = 1.812 \pm 0.014\text{\AA}$ (57) as compared to $1.848 \pm 0.005\text{\AA}$ in $\text{CH}_3\text{SiH}_2\text{F}$ (57) and $r_s = 1.867 \pm$

0.001Å in CH_3SiH_3 (30). A more definite indication of greater polarity in the Ge-F bond as compared to the Si-F bond is given by the CGeF angle of $106^\circ 20'$ in $\text{CH}_3\text{GeH}_2\text{F}$ (64) and the CSiF angle of $109^\circ 13'$ in $\text{CH}_3\text{SiH}_2\text{F}$ (57).

The Ge-Br bond distance has been found to be $r_s = 2.2970 \pm 0.0002\text{\AA}$ in GeH_3Br (67), $r_s = 2.323 \pm 0.001\text{\AA}$ in $(\text{CH}_3)_3\text{GeBr}$ (27), $r_e = 2.277 \pm 0.003\text{\AA}$ in this study, and $r_g = 2.29 \pm 0.02$ in GeBr_4 (29). A redetermination of the molecular structure of GeBr_4 might be expected to place the Ge-Br bond length closer to 2.27Å. It is difficult to make meaningful comparisons of the Si-Br bond length in the corresponding silicon compounds since the distance is known to an accuracy of better than 0.02Å only for BrSiH_3 (68) where a value of $r_s = 2.209 \pm 0.001\text{\AA}$ was obtained. Since (p-d) π interaction of the Ge-F bond in CH_3GeF_3 was found to be small the expected interaction leading to a shortening of the Ge-Br bond in CH_3GeBr_3 would be expected to be negligible. After making corrections (46) for the electronegativity differences the predicted Ge-Br and Si-Br distances are 2.299 and 2.261Å respectively. The agreement is excellent for the Ge-Br bond length in GeH_3Br , but significant shortening is indicated for the Si-Br bond distance in SiH_3Br .

The present study has been unable to accurately determine the Ge-C bond distance in CH_3GeBr_3 due to the relatively small contribution to the molecular scattering curve of atom pairs involving carbon. This distance is expected to be less than the values of $r_s = 1.9400 \pm 0.0001\text{\AA}$ in $(\text{CH}_3)_3\text{GeCl}$ (62) and $r_s = 1.936 \pm 0.006$ in $(\text{CH}_3)_3\text{GeBr}$ (27), but longer than the value $r_e = 1.915 \pm 0.012\text{\AA}$ found in CH_3GeF_3 . The value of 1.87Å

determined for the Ge-C bond distance in CH_3GeBr_3 is very likely too short.

The angle FGeF in CH_3GeF_3 is smaller than the angle BrGeBr in CH_3GeBr_3 , and this would have been predicted from both Bent's (48) and the VSEPR (50) theories. However, non-bonded interactions (49) between the halogen atoms is probably of significance since in both molecules the halogen-halogen non-bonded distance is equal to the Van der Waals radius of interaction.

In conclusion, the structural trends in silicon molecules are closely paralleled by those in germanium compounds. The Ge-C bond length has been shown to have an essentially constant value in the methylgermanes, a feature which has been noted in the methylsilanes (33). Increasing halogen substitution has the effect of decreasing both the Ge-X and Ge-C bond lengths, as is the case in silicon compounds. However, this shortening is much less in germanium structures as compared with silicon molecules. Halogen substitution in a silicon molecule can induce a significant amount of $(p-d)\pi$ bonding, whereas $(p-d)\pi$ overlap in germanium is of much less significance, even in CH_3GeF_3 . It is likely that similar trends will be found in tin and lead structures with $(p-d)\pi$ bonding being negligible. The only comparable study so far reported for these elements is the series $\text{Sn}(\text{CH}_3)_{4-n}\text{Cl}_n$ (69) where a progressive decrease in the

Sn-Cl bond was observed with increasing chlorine substitution.

CHAPTER VII

Suggestions for Further Studies

Our structural knowledge of germanium and silicon compounds contain many gaps, and early determinations frequently have large error estimates. However, some trends can be inferred from those structures that are accurately known and these trends may be tentatively extrapolated to other molecules where our information is incomplete. The discussion which follows is based on r_s bond lengths from microwave studies, and the r_e values estimated from electron diffraction results by equation 1. These are approximately equivalent bond length definitions and are 0.006 - 0.008 Å shorter, depending on the mean square amplitude of vibration, than the usual r_g value derived from electron diffraction.

The contraction of the Si-Cl bond with progressive halogen substitution in the chlorosilane series is, as would be expected, less than that observed for the Si-F bond in the fluorosilanes. The Si-Cl bond length in ClSiH_3 is $2.048 \pm 0.001 \text{ Å}$ (59), and in SiCl_4 it is $2.013 \pm 0.003 \text{ Å}$ (60), a difference of $0.035 \pm 0.004 \text{ Å}$. In the corresponding germanium compounds the Ge-Cl bond length is $2.148 \pm 0.003 \text{ Å}$ in ClGeH_3 (56) and $2.106 \pm 0.001 \text{ Å}$ in GeCl_4 , a difference of $0.042 \pm 0.004 \text{ Å}$. These contractions are fairly similar, and based on the Si-Cl bond length of $2.021 \pm 0.002 \text{ Å}$ in Cl_3SiH (70) one may guess that the Ge-Cl bond length in Cl_3GeH would be about 2.116 Å. Presumably the angle ClGeCl would be the same, or slightly smaller, as the $109^\circ 22' \pm 15'$ for $\angle\text{ClSiCl}$ in Cl_3SiH (70). Although all the fluorosilanes are known with an accuracy of $\pm 0.005 \text{ Å}$ or better (Table V), only monofluorogermane has been studied and the

Ge-F bond length was found to be $1.735 \pm 0.01\text{\AA}$ (65). A determination of the structures of the remaining fluorogermanes and of GeF_4 would be extremely valuable since the possible (p-d) π interaction between germanium and fluorine would be greatest in these molecules. The tetrafluorogermane molecule has been studied previously (28) and the Ge-F bond length was determined as $1.67 \pm 0.03\text{\AA}$. The difference between the Ge-Br bond length in BrGeH_3 and GeBr_4 should be about the same, or slightly less, than the difference in the Ge-Cl bond length in the corresponding chlorine molecules which is 0.042\AA . The only known value for the Ge-Br distance in GeBr_4 is $2.29 \pm 0.02\text{\AA}$ (29) which is longer than this distance in BrGeH_3 where a value of $2.2970 \pm 0.0002\text{\AA}$ was reported (67). Based on this latter determination, the Ge-Br bond length in GeBr_4 should be found in the region of 2.265\AA .

It has been noted that the substitution of hydrogen by a methyl group has the effect of lengthening the M-X bond, where M = Si or Ge and X = halogen. There is some indication that a similar substitution could lengthen the M-M bond since the Si-Si distance in hexamethyldisilane is 2.342 (33) and in disilane it is 2.325\AA (42), although the error involved in $(\text{CH}_3)_3\text{Si}.\text{Si}(\text{CH}_3)_3$ is about 0.01\AA . The Ge-Ge bond length in digermane is $2.397 \pm 0.004\text{\AA}$ (44) and it would be interesting to determine this bond length in hexamethyldigermane.

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PART III

THE CONFORMATION OF RACEMIC
4,6-DIMETHYLTRIMETHYLENE SULFITE

CHAPTER 1

INTRODUCTION

The considerable controversy in recent literature prompted this study of a dimethyl substituted six membered cyclic sulfite in which steric interactions might be expected to play an important role in determining the conformation. The early literature of trimethylene sulfite and its derivatives is mainly concerned with the orientation of the exocyclic oxygen atom (1 - 6), although equilibrium mixtures of conformers are suggested (7,8). The positioning of exocyclic oxygen has since been shown to be axial in solution studies (9 - 13), in X-ray diffraction of solid crystals (14,15) and in the gas phase by electron diffraction (16). The same conclusion has been drawn from infrared studies on trimethylene sulfite in the liquid, solution and gas phases (17).

The first isomers of trimethylene sulfite substituted in the 4 and 6 positions were isolated by Lauterbur (3) who, on the basis of i.r. and n.m.r. spectra assigned the racemic isomer a chair form with equatorial S=O. Overberger (8) concluded from temperature dependance of the n.m.r. and i.r. spectra that a conformational mixture of chair forms (S=O axial and S=O equatorial) was present with S=O axial predominating. However, the proton n.m.r. spectra of the racemic molecule

has been reinterpreted as evidence for non-chair forms and this view is now generally accepted (18). From spectral studies and dipole moments, Wucherpfennig (19) proposed a mixture of twist and perhaps chair forms. A previous spectral study (20) of the analagous racemic 4,6-diisopropyl trimethylene sulfite was in favour of the twist form. Wyn-Jones (21-24) has used ultrasonic techniques to study various sulfites and has concluded that anacomeric equilibria of inverting chair conformations are present. More recently, a study (25) by high resolution n.m.r. spectroscopy of 4,6,6-trimethyl- and 4-phenyl-trimethylene sulfite has indicated that these molecules exist in either a chair-nonchair equilibrium or a twist conformation.

The work of Wood, and co-workers' (26,27,28) has indicated that racemic 4,6-dimethyltrimethylene sulfite is at least partially in a non-chair form in solution, a view that has found some support in the results cited above (19,25).

CHAPTER II

EXPERIMENTAL

A gas chromatographically pure sample of the racemic compound was provided by Dr. M. H. Miskow (27), for which the author is very grateful. Sector electron diffraction photographs were recorded on the Cornell apparatus (29). A suitable vapor pressure was obtained with the sample at a temperature about 10°C lower than that of the nozzle which was at 130°C as measured by an attached thermocouple. Optical densities as a function of radial position were obtained with a digitized microphotometer (30). The data reduction procedure was that of Bauer and his coworkers (30,31), using the Hartley modification to the least square procedure (32) and a weight matrix that includes the first off-diagonal terms (33). Anharmonicity corrections were made for bonded atom pairs (34,35).

CHAPTER III

RESULTS

Experimental relative intensities of four plates were obtained, covering the ranges $q = 14 - 96\text{\AA}^{-1}$ (data set I) and $q = 6 - 93\text{\AA}^{-1}$ (data set II), where $q = (40/\lambda)\sin(\theta/2)$, λ is the electron wavelength and θ the scattering angle. Both sets of data are listed in Appendix H, and the intensities for data set I are plotted with the refined backgrounds in Figure 1.

Six conformations, described using Hilderbrandt's algorithm (36), were tested against the experimental reduced intensity $\frac{\pi}{10} q M_{\text{expt}}(q)$ by the method of least squares. The conformations were labelled A through F and are presented in Figure 2. A larger drawing of model A with numbered atoms is shown in Figure 3. The endocyclic carbon atoms and their attached hydrogen atoms were positioned routinely by means of C-C, C-H and $\angle\text{CCH} = \angle\text{CCC} = \angle\text{HCH}$. The endocyclic oxygens were positioned by means of C-C, C-O, $\angle\text{CCC}$, $\angle\text{CCO}$ and the angle τOCC (e.g. the non-valence angle between C4-C5 and C6-O1. A value of zero for τOCC would correspond to these oxygens lying in the plane formed by C5C6C4). The position of the sulfur atom was obtained using the distance C5...S2 (parameter C....S) and angle H1C5S2 (parameter $\angle\text{SCH}$). This ensures that both S-O lengths are equivalent and conveniently allows the boat or chair conformations to be determined primarily by the value of $\angle\text{SCH}$.

Figure 1 The experimental intensity and background curves for racemic 4,6-dimethyltrimethylene sulfite data set I.

Figure 1

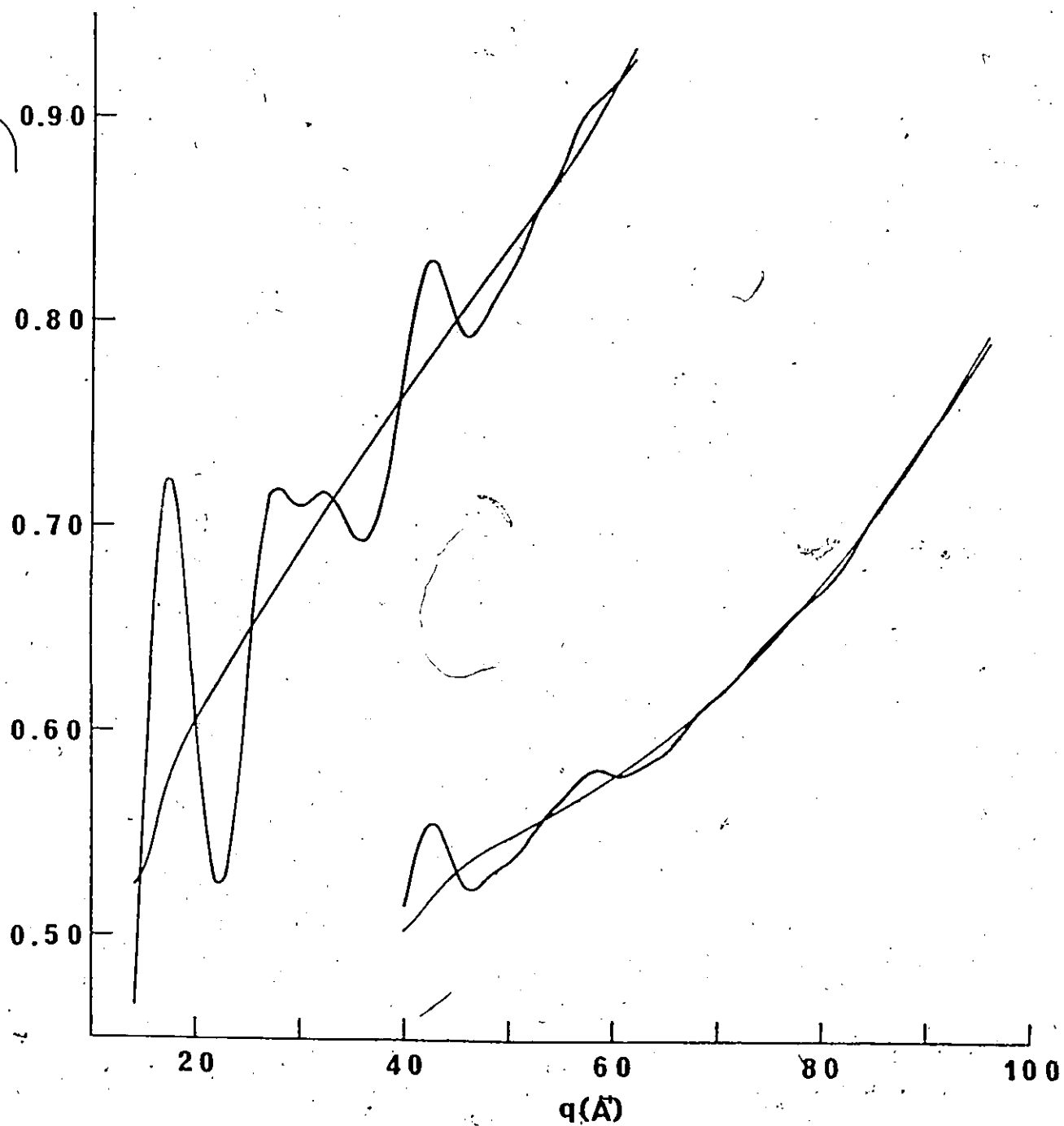


Figure 2 The six different conformations analysed by least squares fitting of the reduced experimental molecular scattering curve.

Figure 2

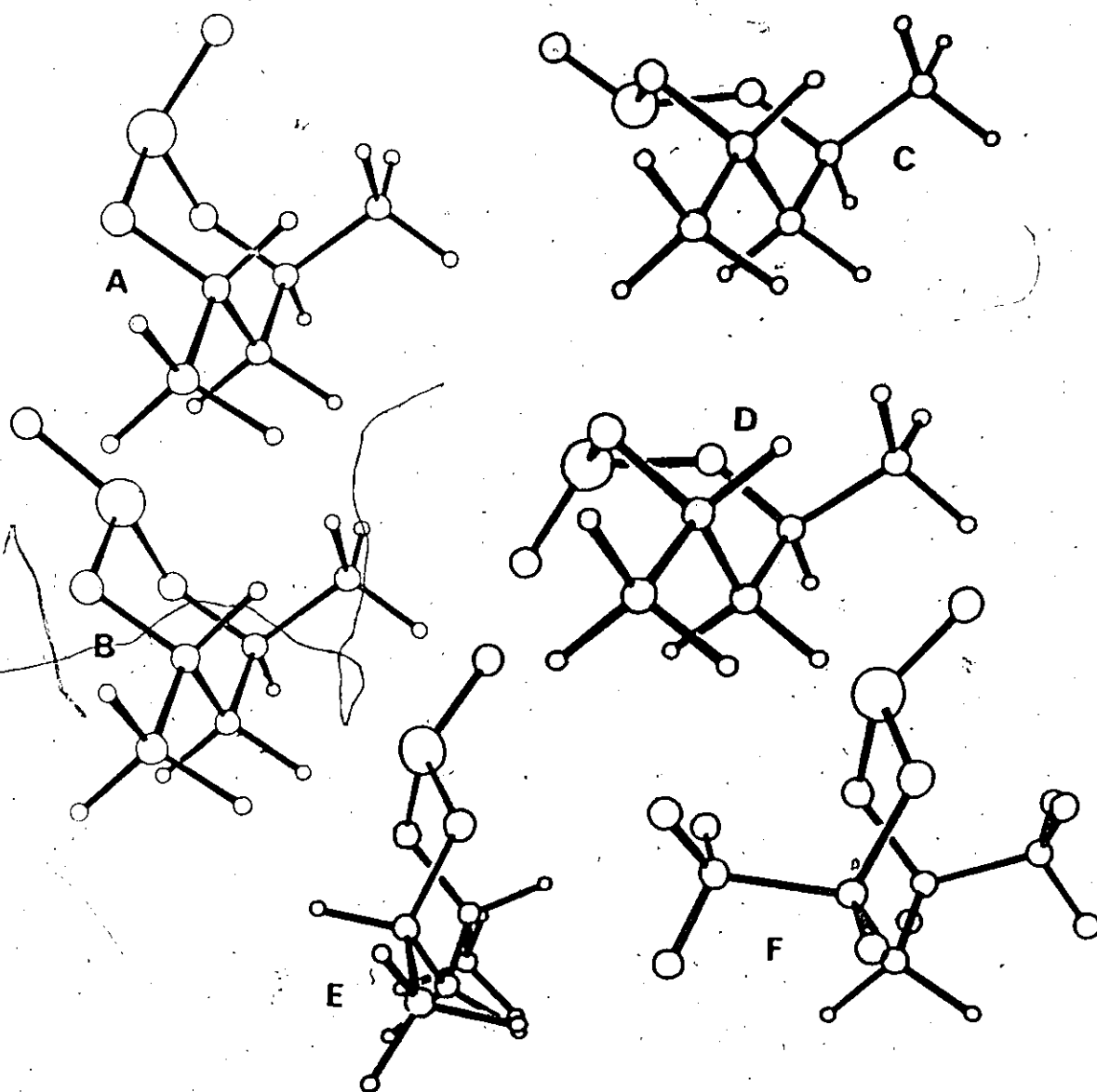
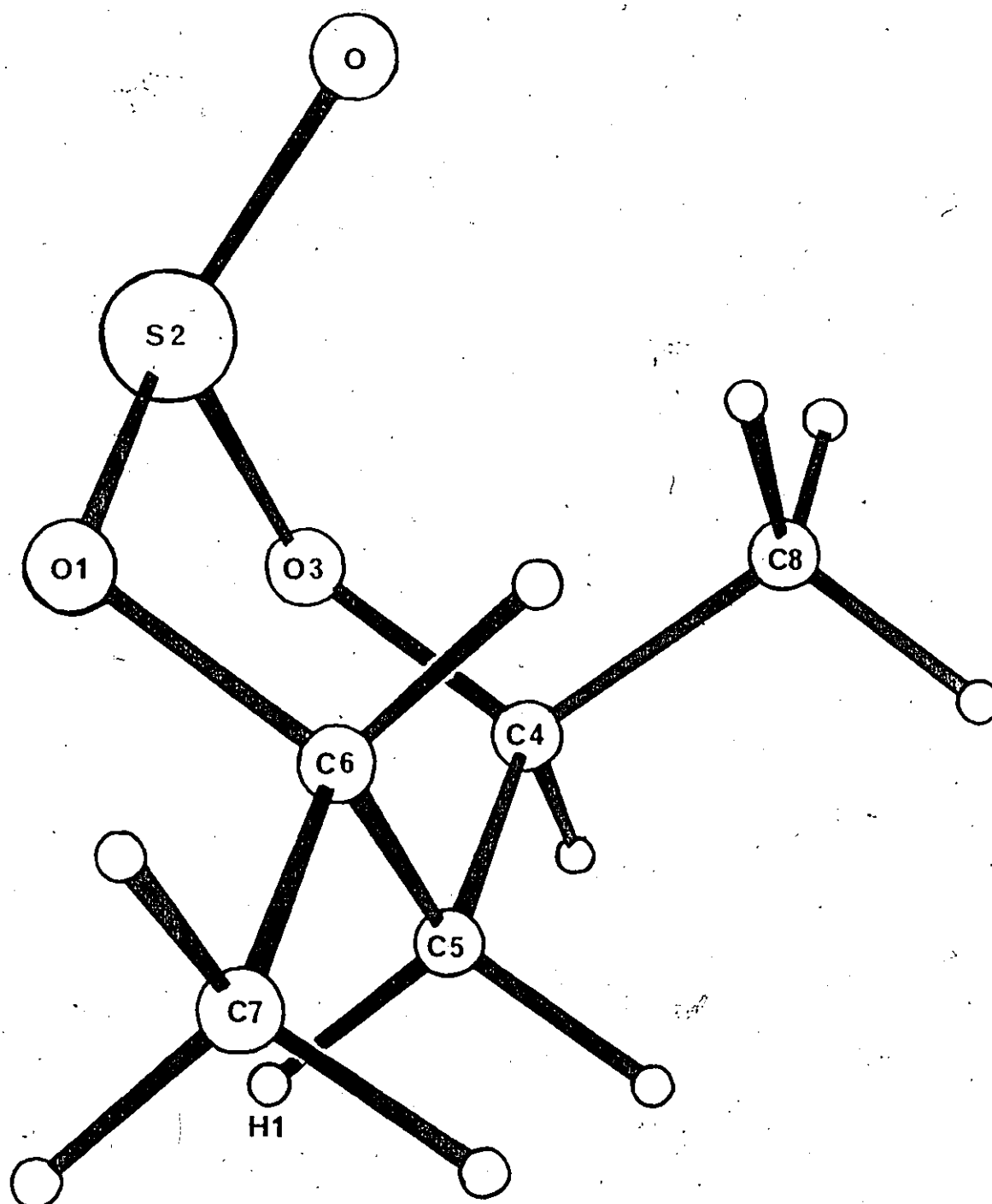


Figure 3. The conformation, with numbered atoms, for model A.

Figure 3



The twist forms were obtained by rotating the C-O bonds against each other by means of τ_{OCC} (See Figure 2). Due to the complexity of the molecule it was assumed that all CCH, HCH and CCC angles were tetrahedral, and that the endo- and exocyclic CCO angles were equivalent.

The selection of initial parameter values for least squares refinement was straight forward for the bond lengths and valence angles. However, initial values of $\angle SCH$, $C\cdots S$ and τ_{OCC} had to be found by trial and error in order that the redundant parameters $\angle COS$, $\angle OSO$ and $S-O$ would have reasonable starting values, as calculated from the interatomic distance matrix. In order to test each conformation it was necessary to attempt a refinement of as many parameters as possible, although under these conditions large correlation coefficients were found between the parameters and convergence was difficult to obtain.

Radial distribution curves were calculated using a value of 0.0025 for the Degard damping constant. Only the contributions due to C-H and S-O were clearly resolved. By fitting the intensity curves we find two solutions for each of the chair models, placing the S-O distance at 1.62Å. The parameter values obtained when both sets of data were treated independently for the preferred chair form of model A are given in Table I. The corresponding interatomic distances and mean square amplitudes of vibration for the heavy atoms are given in Table II and the correlation coefficients between the parameters in this solution are listed in Table III. Some of the parameters are highly correlated,

Table I

The Structural Parameters¹ and Standard Deviations Obtained from Independent Least Squares Fits to Data Set I ($q = 14 - 96 \text{ \AA}^{-1}$) and Data Set II ($q = 6 - 93 \text{ \AA}^{-1}$) for Model A

Parameter	Data Set I		Data Set II		Weighted Value ²
	Value	σ	Value	σ	
C-H	1.125	0.0058	1.106	0.0090	1.119
C-C	1.505	0.0044	1.505	0.0049	1.505
C-O	1.411	0.0035	1.417	0.0058	1.413
S-O	1.479	0.0068	1.481	0.0081	1.480
<CCO	111.2	0.3956	110.9	0.5199	111.1
<OS=O	116.8	0.3942	116.4	0.6247	116.7
C...S	3.002	0.0185	3.011	0.0106	3.009
<SCH	101.8	0.7508	101.2	0.2356	100.3
<OCC	58.8	0.9246	59.8	0.2847	59.7
Redundant Parameters					
S-O	1.621	0.0032	1.623	0.0026	1.622
<COS	114.0	0.2481	114.9	0.5894	114.1
<OSO	100.7	0.5815	100.8	0.8304	100.7
R ³	0.0646		0.1010		

¹ Bond lengths are r_g values in \AA , angles are in degrees. The following angles were constrained to be tetrahedral: <CCC, <CCH, <HCH.

Table 1 Continued

2. The values are weighted by the standard deviations:

$$X = \frac{\sum_i (x_i / \sigma_i^2)}{\sum_i (1 / \sigma_i^2)}$$

$$3_R = \left[\frac{\sum_q \left(\frac{\pi}{10} q M(q)_{\text{expt}} - \frac{\pi}{10} q M(q)_{\text{theor}} \right)^2 \cdot w_q}{\sum_q \left(\frac{\pi}{10} q M(q)_{\text{expt}} \right)^2 \cdot w_q} \right]^{1/2}$$

Table II

	<u>0</u>	<u>01</u>	<u>S2</u>	<u>03</u>	<u>C4</u>	<u>C5</u>	<u>C6</u>	<u>C7</u>	<u>C8</u>
0	0.0	2.642 (0.060)	1.479 (0.047)	2.642 (0.060)	3.247 (0.121)	3.875 (0.121)	3.247 (0.121)	4.587 (0.121)	2.965 (0.121)
01		0.0	1.621 (0.049)	2.497 (0.070)	2.850 (0.121)	2.406 (0.075)	1.411 (0.050)	2.406 (0.075)	3.493 (0.121)
S2			0.0	1.621 (0.049)	2.546 (0.083)	3.002 (0.121)	2.546 (0.083)	3.892 (0.121)	2.996 (0.121)
03				0.0	1.411 (0.050)	2.406 (0.075)	2.850 (0.121)	4.250 (0.121)	2.406 (0.075)
C4					0.0	1.505 (0.050)	2.455 (0.082)	3.786 (0.121)	1.505 (0.050)
C5						0.0	1.505 (0.050)	2.458 (0.082)	2.458 (0.082)
C6							0.0	1.505 (0.050)	2.928 (0.121)
C7								0.0	4.273 (0.121)
C8									0.0

1 These are r_g values, in Å.

2 The $11j$ values presented in this table were determined during previous least squares analysis and were constrained during the later stages of refinement. Mean square amplitudes involving hydrogen were as follows: C-H = 0.080; H...H = 0.120; with geminal C or O atom = 0.100; with all other heavy atoms = 0.150; all long H...H distances = 0.200. All values are in Å.

3 The conformation and designation of the atoms corresponds to Figure 3. The distances presented above were calculated for this conformation using the parameter values listed for data set I in Table I.

Table III

Diagonal Elements and Correlation Coefficients for Model A¹

	C-H	C-C	C-O	<CCO	τ CC	C...S	<SCH	S=O	<OS=O
σ	0.0058	0.0044	0.0035	0.3956	0.9246	0.0185	0.7508	0.0068	0.3942
C-H	1	-0.01	0.05	0.10	-0.02	0.00	-0.01	0.04	-0.02
C-C		1	0.36	-0.86	-0.70	0.77	-0.69	-0.89	0.53
C-O			1	-0.51	-0.25	0.41	-0.26	-0.44	0.30
<CCO				1	0.60	-0.78	0.58	0.84	-0.37
τ CC					1	-0.94	0.98	0.69	-0.20
C...S						1	-0.90	-0.74	0.17
<SCH							1	0.58	-0.24
S=O								1	-0.64
<OS=O									1

¹The individual standard deviations are given by

$$\sigma = \left[\frac{B_{ii}^{-1} V' W W'}{n-m} \right]^{1/2}$$

, and the correlation coefficients by

$$\rho_{ij} = \frac{B_{ij}^{-1}}{(B_{ii}^{-1} B_{jj}^{-1})^{1/2}}$$

and although convergence was not obtained, the parameter values were oscillating with amplitudes less than the given uncertainties.

The alternate form of model A, again refined independently from the two sets of data, gave valence parameter values in close agreement with each other, and within the errors given for the preferred model, with the exceptions of $\angle OS=0$, $\angle COS$ and $\angle OSO$. For these respective parameters were determined the values 107.9° , 117.9° , and 106.9° from data set I ($R = 0.0846$) and 108.6° , 118.3° and 106.1° from data set II ($R = 0.0986$). Although the fit to data set II for this form is better than that given in Table I, it is not significantly so at the 99.5% confidence level (37,38), and the values for $\angle OS=0$ are smaller than expected.

A similar result was obtained for model B, the chair form with equatorial S=O bond, although the best fits to both data sets were obtained with parameter values similar to those discussed above for the alternate form of model A. These results are presented in Table IV. On comparing Tables I and IV it is evident that the differences in the chair forms are due largely to the parameters $\angle SCH$ and τOCC . Transferring the parameters given in Table I as an initial structure for model B, we find that allowing the valence and conformational determining parameters to vary results in solutions with valence parameters as given in Table I, within the given errors, with the exception of $\angle OSO$. Data sets I and II yield values of 92.4° and 91.0° respectively for $\angle OSO$, values which are approximately 10° less than the similar angle determined in trimethylene

Table IV

The Structural Parameters and Standard Deviations Obtained from Independent Least Square Fits to Data Set I and Data Set II for Model B.

Parameter	Data Set I		Data Set II		Weighted Value
	Value	σ	Value	σ	
C-H	1.117	0.0075	1.101	0.0083	1.110
C-C	1.512	0.0036	1.514	0.0039	1.513
C-O	1.427	0.0060	1.428	0.0062	1.427
S-O	1.469	0.0070	1.473	0.0078	1.471
<CCO	108.8	0.4288	109.1	0.4726	108.9
<OS=O	108.3	0.4747	109.3	0.5645	108.7
C...S	2.983	0.0120	2.992	0.0139	2.987
<SCH	89.9	0.7228	88.9	0.8770	89.5
τ OCC	66.1	0.7332	65.9	0.8805	66.0
Redundant Parameters					
S-O	1.621	0.0024	1.626	0.0028	1.623
<COS	119.3	0.5833	120.3	0.6766	119.7
<OSO	105.8	0.8430	106.0	0.9095	105.9
R	0.0746			0.0863	

sulfite (14,16) and 2,2 -dichloro trimethylene sulfite (15).

The boat forms C and D were analysed using data set I only. The best fits obtained for these conformations had R values of 0.1878 and 0.2186 respectively, and are not significant at the 99.5% confidence level (37,38). Partially twisted boat conformations did not successfully refine, hence the completely twisted forms E and F were assumed by constraining the sulfur atom to the position shown in Figure 2 by means of $\angle SCH = 125.25^\circ$. Model E was analysed using data set I only and refined to a set of valence parameter values consistent with those of Table I, but again the fit ($R = 0.1701$) is not statistically significant. Model F was analysed using both data sets, but the fits obtained ($R = 0.1101$ for Set I and $R = 0.1469$ for set II) are still not significant at the 99.5% confidence level (37,38).

The experimental reduced intensity $\frac{\pi}{10} q M_{\text{expt}}(q)$ is compared with $\frac{\pi}{10} q M_{\text{theor.}}(q)$ for A in Figure 4 along with the difference curves for A through F. The corresponding radial distribution curves calculated from the reduced intensity curves are presented in Figure 5 for the region of structural interest. The full experimental radial distribution curve is compared with that calculated for model A in Figure 6.

Figure 4 The reduced experimental molecular scattering curve, the corresponding theoretical curve calculated for model A, and the difference curves (experimental - theoretical) for models A through F data set I.

Figure 4

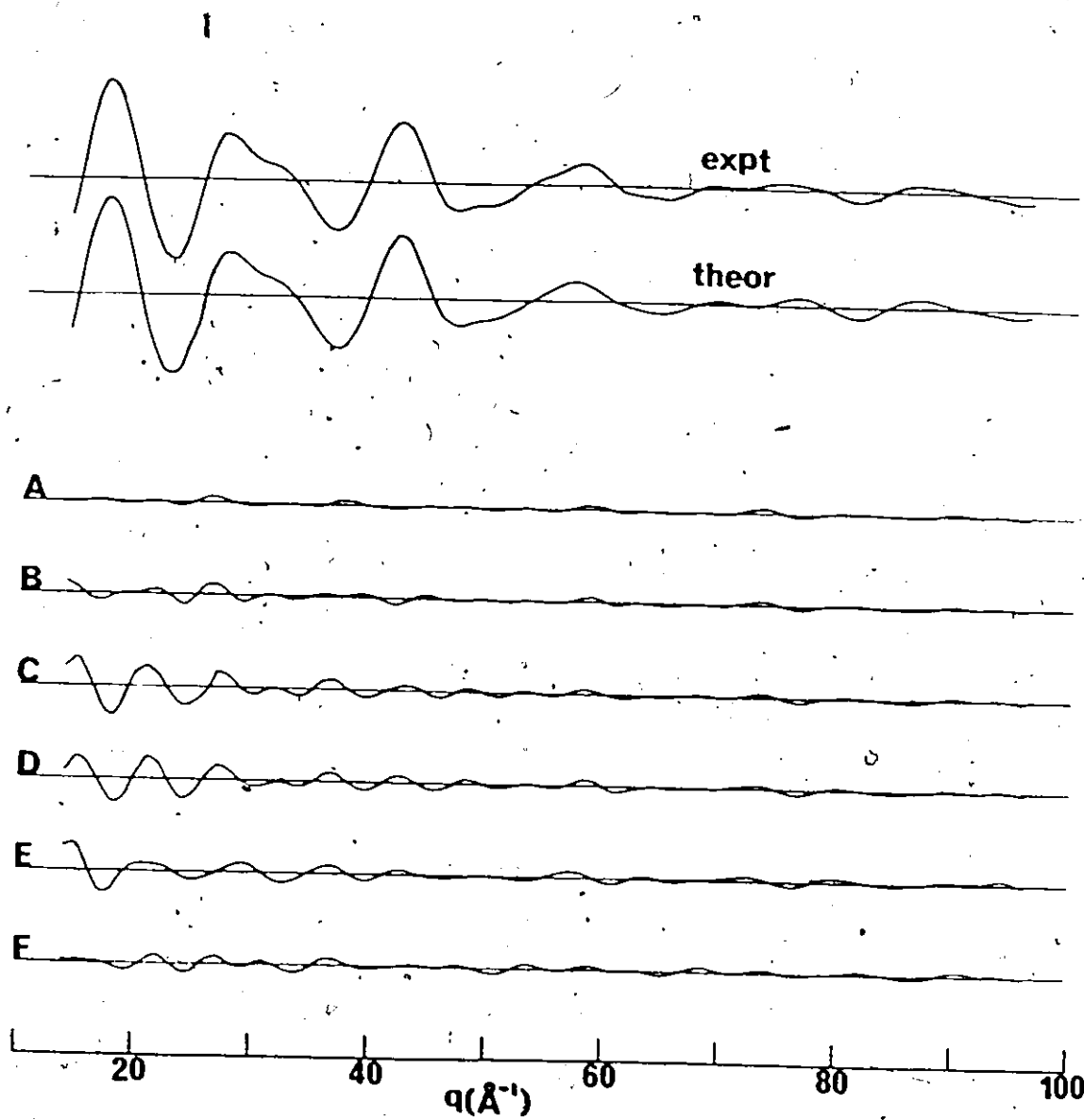


Figure 5 The experimental radial distribution function in the region of structural interest, together with the corresponding theoretical function for model A and the difference curves (experimental - theoretical) for models A through F.

Figure 5

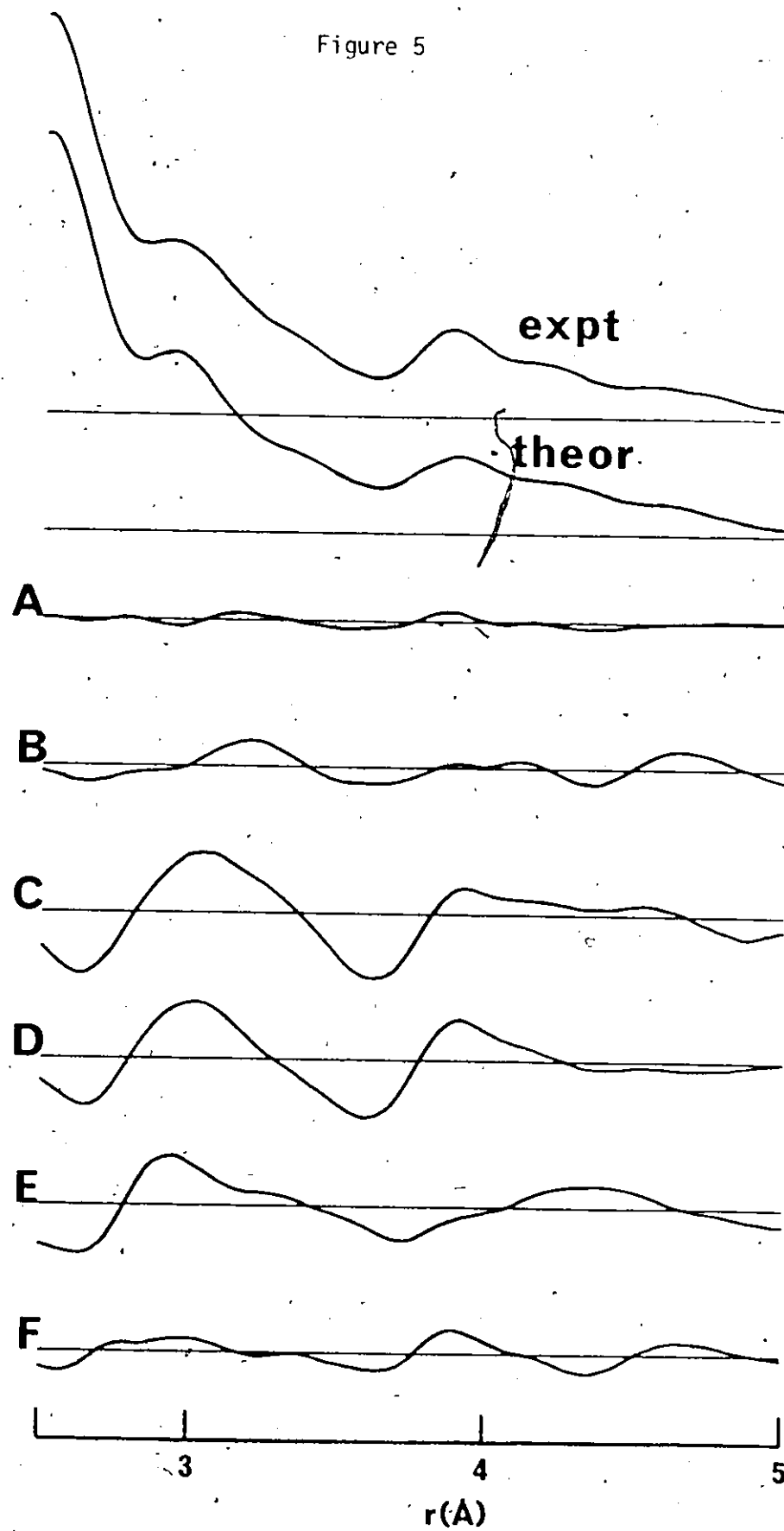
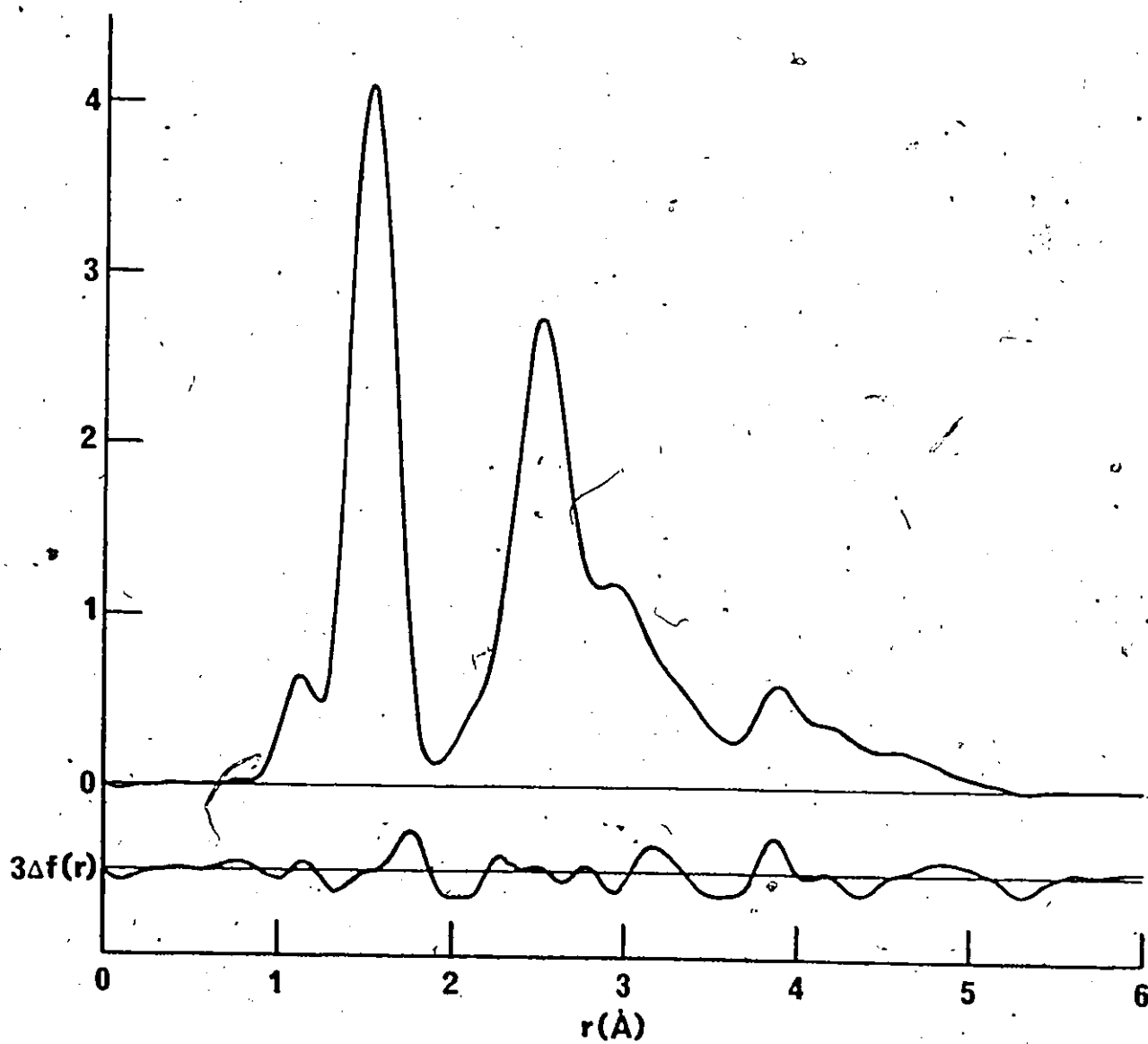


Figure 6 The refined experimental radial distribution curve.

$$\Delta f(r) = [f(r)_{\text{expt}} - f(r)_{\text{theor}}]$$

Figure 6



CHAPTER IV

DISCUSSION

While the analyses of the two sets of data are consistent in that the parameter values obtained for models A, B and F are the same, within calculated errors, the normalized standard deviations for models A and B are not consistent: for data set I the lowest value for R is given by model A, the chair form with axial S=O bond, while for data set II the lowest value for R is given by model B, a chair form having different ring valence angles than those of model A, with an equatorial bond. However, the R factor ratio $0.0746/0.0646 = 1.155$ for the two fits to data set I does not allow us to statistically reject the hypothesis that model B is an acceptable solution at the 1% level of significance for this data (37,38).

We tested the tetrahedral constraint on $\angle CCC$ for model A and data set I. Using the parameters of Table I as an initial model, we found a minimum ($R = 0.0585$) with the exocyclic $\angle CCC = 111.1^\circ$ and the endocyclic angle $= 106.6^\circ$. As a second test we constrained $\angle CCC$ at 110.5° and obtained a fit with $R = 0.0704$. In both cases the other parameters were unchanged, within the estimated errors, and the addition of $\angle CCC$ to the varied parameters, or the change in the value of the constraint, do not make a significant difference to the fit obtained. We therefore conclude that $\angle CCC$ might well be a degree or so larger than tetrahedral.

A more serious constraint of our analysis was the use of the same mean square amplitudes of vibration for all the models. The mean square amplitudes given in Table II are certainly not unique, and we would expect different results depending on the values used. However, it is beyond the capabilities of our experiment to determine the non-bonded amplitudes for this molecule, so we have assumed a reasonable set of values and admit that an alternate set might give an improved fit to any of the models tested, although it is highly unlikely that the rejected models would become statistically significant.

The parameter values for models A and B are again listed and compared with the values in trimethylene sulfite (14,16) and 2,2-dichlorotrimethylene sulfite (15) in Table V. The errors given for the two models in the present work are three times the calculated uncertainties and are estimated to be at the 95% confidence level and to include systematic errors. It can be seen that model A agrees reasonably well with the electron diffraction results for the unsubstituted molecule, while model B compares more favourably with the X-ray results, particularly in that $\langle \cos \rangle$ is much larger than $\langle \cos \rangle = 0$. This is an unexpected trend, and we note that in model A the peak in the radial distribution curve just less than 4 Å is due to the exocyclic oxygen to carbon distances. When this oxygen is placed in the equatorial position, as in model B, it is precisely these distances which change the most. It is therefore reasonable to assume, as Naumov

Table V

Comparison of the molecular parameters¹ for racemic 4,6-dimethyltrimethylene sulfite with trimethylene sulfite (TMS) and 2,2-dichlorotrimethylene sulfite (DTS)

Parameters	Model A	Model B	E.D. TMS (16)	X-Ray TMS (14)	X-ray DTS (15)
C-H	$1.119 \pm 0.020(0.08)$	1.110 ± 0.023	$1.12(\text{assumed})(0.08)$		
C-C	$1.505 \pm 0.014(0.05)$	1.513 ± 0.011	$1.52 \pm 0.02(0.049)$	1.52 ± 0.018	1.523 ± 0.008
C-O	$1.413 \pm 0.013(0.05)$	1.427 ± 0.018	$1.42 \pm 0.02(0.049)$	1.46 ± 0.015	1.462 ± 0.007
S-O	$1.480 \pm 0.022(0.05)$	1.471 ± 0.022	$1.45 \pm 0.02(0.049)$	1.45 ± 0.011	1.442 ± 0.006
S-O	$1.622 \pm 0.009(0.05)$	1.623 ± 0.007	$1.62 \pm 0.01(0.055)$	1.60 ± 0.008	1.622 ± 0.004
<CCC	$109.5(\text{assumed})$	$109.5(\text{assumed})$	110 ± 3	110 ± 1.3	110.6 ± 0.6
<CCO	111.1 ± 1.3	108.9 ± 1.3	112 ± 3	109 ± 1.2	108.8 ± 0.4
<OSO	100.7 ± 2.0	105.9 ± 2.6	99 ± 2	100 ± 0.6	97.8 ± 0.3
<COS	114.1 ± 1.0	119.7 ± 1.9	113 ± 2	116 ± 0.7	116.2 ± 0.3
<OS=O	116.7 ± 1.4	108.7 ± 1.5	113.5 ± 3	107 ± 0.4	107.2 ± 0.2

¹ Bond lengths in Å, angles in degrees. The mean square amplitudes of vibration are given in parentheses.

and his co-workers have pointed out (15), that the $\angle OS=O$ would tend to become much smaller than that determined for an axial $S=O$ model in an effort to make these distances the same as before.

In summary, the electron diffraction data is not consistent with a static boat, or fully twisted conformation for racemic 4,6-dimethyl trimethylene sulfite. It has been pointed out (25) that the axial position for the $S=O$ group experiences a minimum of gauche interactions with the geminal electron pairs at the adjacent ring atoms, as compared to an equatorial $S=O$ bond. The available evidence favors a chair form with an axial $S=O$ bond, although the equatorial form, or mixture of conformers could not be definitely eliminated.

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APPENDIX A

SECTOR FUNCTION FROM CARBON DISULFIDE SCATTERING

LONG DISTANCE 296.16MM

7	0.00426	8	0.00500	9	0.00573	10	0.00660	11	0.00796
12	0.01057	13	0.01505	14	0.02181	15	0.03108	16	0.04207
17	0.05427	18	0.06648	19	0.07887	20	0.09241	21	0.10739
22	0.12455	23	0.14437	24	0.16662	25	0.19316	26	0.22374
27	0.25967	28	0.30061	29	0.34443	30	0.39286	31	0.44302
32	0.49718	33	0.55756	34	0.62525	35	0.69739	36	0.77498
37	0.85780	38	0.94668	39	1.04446	40	1.15603	41	1.27700
42	1.40752	43	1.55184	44	1.69151	45	1.82883		

SHORT DISTANCE 95.45MM

35	0.01044	36	0.01126	37	0.01228	38	0.01350	39	0.01498
40	0.01682	41	0.01892	42	0.02132	43	0.02411	44	0.02732
45	0.03092	46	0.03478	47	0.03896	48	0.04347	49	0.04832
50	0.05346	51	0.05873	52	0.06416	53	0.06983	54	0.07550
55	0.08125	56	0.08700	57	0.09278	58	0.09859	59	0.10438
60	0.11025	61	0.11620	62	0.12244	63	0.12871	64	0.13512
65	0.14163	66	0.14819	67	0.15494	68	0.16197	69	0.16872
70	0.17587	71	0.18336	72	0.19120	73	0.19926	74	0.20773
75	0.21676	76	0.22601	77	0.23565	78	0.24539	79	0.25532
80	0.26688	81	0.27804	82	0.28978	83	0.30166	84	0.31406
85	0.32713	86	0.34083	87	0.35521	88	0.37017	89	0.38527
90	0.40077	91	0.41667	92	0.43288	93	0.44958	94	0.46668
95	0.48398	96	0.50204	97	0.51961	98	0.53792	99	0.55735
100	0.57739	101	0.59738	102	0.61836	103	0.63990	104	0.66134
105	0.68493	106	0.70916	107	0.73361	108	0.75710	109	0.78056
110	0.80475	111	0.82991	112	0.85585	113	0.88288	114	0.90975
115	0.93671	116	0.96529	117	0.99493	118	1.02673	119	1.05966
120	1.09213	121	1.12398	122	1.15652	123	1.19048	124	1.22477
125	1.26006	126	1.29668	127	1.33362	128	1.36902	129	1.40768
130	1.44795	131	1.48912	132	1.53396	133	1.57875	134	1.62359
135	1.67082								

Appendix B

Calculation of Scale Parameters from Magnesium Oxide

Approximate MgO-photographic Plate Distance = 307.05 mm

Miller Index	Diameter Measurements mm			Average Radius mm	Standard Deviation	$s(\text{experimental})^1$	$s(\text{theoretical})^2$
111	12.605	12.625	12.600	12.615	6.306	0.005	0.204985
200	14.594	14.619	14.611	14.610	7.304	0.005	0.236697
220	20.667	20.655	20.715	20.686	10.340	0.011	0.334740
222	25.351	25.315	25.347	25.353	12.671	0.008	0.409971
400	29.276	29.266	29.290	29.304	14.642	0.007	0.473394
420	32.703	32.723	32.765	32.754	16.368	0.012	0.529270
422	35.862	35.819	35.852	35.885	17.927	0.012	0.579786
440	41.396	41.442	41.442	41.417	20.712	0.010	0.669480
600	43.941	43.945	43.969	44.011	21.983	0.014	0.710090
620	46.362	46.336	46.436	46.448	23.198	0.024	0.748501

Final scale parameters were $L = 307.24$ mm

$$\lambda = 0.050279 \text{ \AA}$$

$$^1 s(\text{experimental}) = \left\{ (4\pi/\lambda) \sin(0.5 \tan^{-1}(r/L)) \right\}$$

$$^2 s(\text{theoretical}) = 4\pi (h^2 + k^2 + l^2)^{1/2} / a$$

APPENDIX D

EXPERIMENTAL INTENSITIES FOR (CH₃)₄GF

CURVE I. WAVELENGTH=0.05060A SAMPLE DISTANCE=296.16MM									
7 0.7459	8 0.5775	9 0.4548	10 0.3736	11 0.3531	12 0.3712	13 0.4135	14 0.4576	15 0.4831	16 0.4855
17 0.4865	18 0.5012	19 0.5294	20 0.5557	21 0.5668	22 0.5691	23 0.5696	24 0.5671	25 0.5586	26 0.5386
27 0.5127	28 0.4925	29 0.4882	30 0.4977	31 0.5173	32 0.5332	33 0.5384	34 0.5338	35 0.5161	36 0.4971
37 0.4859	38 0.4796	39 0.4811	40 0.4846	41 0.4881	42 0.4898	43 0.4890			
CURVE II. WAVELENGTH=0.05060A SAMPLE DISTANCE=296.16MM									
7 0.7581	8 0.5784	9 0.4482	10 0.3743	11 0.3493	12 0.3676	13 0.4121	14 0.4562	15 0.4812	16 0.4829
17 0.4847	18 0.5007	19 0.5281	20 0.5529	21 0.5646	22 0.5660	23 0.5655	24 0.5627	25 0.5537	26 0.5330
27 0.5082	28 0.4891	29 0.4853	30 0.4959	31 0.5135	32 0.5302	33 0.5350	34 0.5272	35 0.5092	36 0.4925
37 0.4805	38 0.4756	39 0.4768	40 0.4796	41 0.4830	42 0.4844	43 0.4836			
CURVE III. WAVELENGTH=0.05054A SAMPLE DISTANCE=95.45MM									
35 0.3355	36 0.3137	37 0.3011	38 0.2982	39 0.3024	40 0.3123	41 0.3228	42 0.3335	43 0.3451	44 0.3554
45 0.3630	46 0.3685	47 0.3745	48 0.3830	49 0.3926	50 0.4021	51 0.4109	52 0.4191	53 0.4237	54 0.4243
55 0.4215	56 0.4159	57 0.4102	58 0.4049	59 0.4002	60 0.3967	61 0.3933	62 0.3901	63 0.3856	64 0.3795
65 0.3719	66 0.3637	67 0.3563	68 0.3505	69 0.3431	70 0.3415	71 0.3410	72 0.3371	73 0.3329	74 0.3287
75 0.3228	76 0.3185	77 0.3144	78 0.3110	79 0.3079	80 0.3060	81 0.3047	82 0.3034	83 0.3021	84 0.3005
85 0.2991	86 0.2972	87 0.2955	88 0.2941	89 0.2927	90 0.2917	91 0.2909	92 0.2901	93 0.2895	94 0.2879
95 0.2861	96 0.2843	97 0.2828	98 0.2810	99 0.2797	100 0.2784	101 0.2774	102 0.2763	103 0.2747	104 0.2737
105 0.2726	106 0.2711	107 0.2694	108 0.2680	109 0.2666	110 0.2652	111 0.2643			
CURVE IV. WAVELENGTH=0.05054A SAMPLE DISTANCE=95.45MM									
35 0.4868	36 0.4554	37 0.4389	38 0.4350	39 0.4440	40 0.4593	41 0.4760	42 0.4937	43 0.5118	44 0.5281
45 0.5393	46 0.5485	47 0.5571	48 0.5677	49 0.5833	50 0.6004	51 0.6136	52 0.6237	53 0.6305	54 0.6309
55 0.6257	56 0.6183	57 0.6097	58 0.6014	59 0.5953	60 0.5899	61 0.5858	62 0.5816	63 0.5750	64 0.5669
65 0.5576	66 0.5456	67 0.5343	68 0.5247	69 0.5179	70 0.5128	71 0.5091	72 0.5053	73 0.5011	74 0.4954
75 0.4890	76 0.4828	77 0.4771	78 0.4722	79 0.4682	80 0.4657	81 0.4638	82 0.4622	83 0.4610	84 0.4592
85 0.4567	86 0.4544	87 0.4518	88 0.4498	89 0.4486	90 0.4479	91 0.4468	92 0.4458	93 0.4445	94 0.4430
95 0.4404	96 0.4379	97 0.4357	98 0.4333	99 0.4312	100 0.4291	101 0.4272	102 0.4261	103 0.4245	104 0.4225
105 0.4208	106 0.4189	107 0.4166	108 0.4143	109 0.4123	110 0.4106	111 0.4083			
CURVE V. WAVELENGTH=0.05054A SAMPLE DISTANCE=95.45MM									
35 0.4889	36 0.4562	37 0.4376	38 0.4336	39 0.4418	40 0.4568	41 0.4741	42 0.4909	43 0.5083	44 0.5244
45 0.5364	46 0.5451	47 0.5534	48 0.5645	49 0.5805	50 0.5977	51 0.6139	52 0.6267	53 0.6336	54 0.6331
55 0.6285	56 0.6220	57 0.6161	58 0.6086	59 0.6021	60 0.5970	61 0.5928	62 0.5883	63 0.5812	64 0.5712
65 0.5591	66 0.5462	67 0.5349	68 0.5255	69 0.5166	70 0.5107	71 0.5068	72 0.5028	73 0.4979	74 0.4921
75 0.4863	76 0.4804	77 0.4741	78 0.4690	79 0.4653	80 0.4629	81 0.4607	82 0.4590	83 0.4575	84 0.4550
85 0.4526	86 0.4497	87 0.4472	88 0.4456	89 0.4437	90 0.4422	91 0.4414	92 0.4404	93 0.4387	94 0.4369
95 0.4348	96 0.4325	97 0.4300	98 0.4275	99 0.4252	100 0.4234	101 0.4219	102 0.4203	103 0.4187	104 0.4169
105 0.4146	106 0.4129	107 0.4106	108 0.4082	109 0.4063	110 0.4045	111 0.4027			

APPENDIX E

EXPERIMENTAL INTENSITIES FOR CH3GEF3

CURVE I. WAVELENGTH=0.05007 SAMPLE DISTANCE=296.16MM															
7	0.4707	8	0.3556	9	0.2821	10	0.2373	11	0.2233	12	0.2430	13	0.2911	14	0.3566
17	0.4399	18	0.4169	19	0.3935	20	0.3847	21	0.3925	22	0.4113	23	0.4320	24	0.4471
27	0.4490	28	0.4381	29	0.4269	30	0.4124	31	0.4037	32	0.4000	33	0.4011	34	0.4037
37	0.4139	38	0.4129	39	0.4078	40	0.3997	41	0.3897	42	0.3804	43	0.3738	44	0.3704
45												45	0.3707		
CURVE II. WAVELENGTH=0.05007 SAMPLE DISTANCE=296.16MM															
7	0.4772	8	0.3545	9	0.2779	10	0.2339	11	0.2210	12	0.2420	13	0.2934	14	0.3596
17	0.4402	18	0.4168	19	0.3931	20	0.3852	21	0.3942	22	0.4132	23	0.4332	24	0.4482
27	0.4510	28	0.4389	29	0.4238	30	0.4111	31	0.4031	32	0.4007	33	0.4004	34	0.4033
37	0.4134	38	0.4121	39	0.4070	40	0.3986	41	0.3884	42	0.3794	43	0.3777	44	0.3694
45												45	0.3694		
CURVE III. WAVELENGTH=0.05007 SAMPLE DISTANCE=296.16MM															
7	0.5726	8	0.4235	9	0.3226	10	0.2727	11	0.2621	12	0.2904	13	0.3521	14	0.4279
17	0.5137	18	0.4840	19	0.4578	20	0.4494	21	0.4601	22	0.4835	23	0.5065	24	0.5242
27	0.5258	28	0.5121	29	0.4964	30	0.4819	31	0.4716	32	0.4668	33	0.4674	34	0.4713
37	0.4847	38	0.4834	39	0.4779	40	0.4672	41	0.4581	42	0.4450	43	0.4364	44	0.4316
45												45	0.4316		
CURVE IV. WAVELENGTH=0.05007 SAMPLE DISTANCE=296.16MM															
7	0.5655	8	0.4249	9	0.3356	10	0.2816	11	0.2611	12	0.2817	13	0.3353	14	0.4078
17	0.5165	18	0.4908	19	0.4622	20	0.4489	21	0.4563	22	0.4772	23	0.5018	24	0.5205
27	0.5275	28	0.5161	29	0.5003	30	0.4851	31	0.4737	32	0.4675	33	0.4665	34	0.4692
37	0.4834	38	0.4834	39	0.4788	40	0.4694	41	0.4582	42	0.4462	43	0.4372	44	0.4315
45												45	0.4298		
CURVE V. WAVELENGTH=0.05015A SAMPLE DISTANCE=95.45MM															
35	0.3148	36	0.3071	37	0.3031	38	0.3001	39	0.2992	40	0.2990	41	0.2987	42	0.3002
45	0.3237	46	0.3180	47	0.3104	48	0.3013	49	0.3099	50	0.3159	51	0.3196	52	0.3205
55	0.3379	56	0.3355	57	0.3341	58	0.3335	59	0.3334	60	0.3327	61	0.3347	62	0.3350
65	0.3462	66	0.3397	67	0.3346	68	0.3302	69	0.3260	70	0.3232	71	0.3211	72	0.3193
75	0.3102	76	0.3064	77	0.3028	78	0.2992	79	0.2961	80	0.2951	81	0.2944	82	0.2939
85	0.2915	86	0.2904	87	0.2890	88	0.2876	89	0.2861	90	0.2844	91	0.2835	92	0.2829
95	0.2823	96	0.2815	97	0.2804	98	0.2791	99	0.2774	100	0.2760	101	0.2745	102	0.2734
105	0.2712	106	0.2707	107	0.2701	108	0.2690	109	0.2680	110	0.2668	111	0.2658	112	0.2645
113												113	0.2631	114	0.2622
CURVE VI. WAVELENGTH=0.05015A SAMPLE DISTANCE=95.45MM															
35	0.3192	36	0.3121	37	0.3077	38	0.3048	39	0.3040	40	0.3034	41	0.3034	42	0.3046
45	0.3291	46	0.3228	47	0.3150	48	0.3059	49	0.3145	50	0.3207	51	0.3241	52	0.3264
55	0.3325	56	0.3298	57	0.3286	58	0.3276	59	0.3275	60	0.3270	61	0.3275	62	0.3296
65	0.3306	66	0.3241	67	0.3187	68	0.3137	69	0.3095	70	0.3273	71	0.3253	72	0.3229
75	0.3137	76	0.3098	77	0.3062	78	0.3029	79	0.3005	80	0.2986	81	0.2960	82	0.2977
85	0.2951	86	0.2936	87	0.2923	88	0.2908	89	0.2891	90	0.2880	91	0.2879	92	0.2869
95	0.2858	96	0.2855	97	0.2845	98	0.2829	99	0.2814	100	0.2795	101	0.2781	102	0.2776
105	0.2750	106	0.2745	107	0.2738	108	0.2728	109	0.2722	110	0.2709	111	0.2704	112	0.2680
113												113	0.2675	114	0.2670
CURVE VII. WAVELENGTH=0.05015A SAMPLE DISTANCE=95.45MM															
35	0.3112	36	0.3037	37	0.3006	38	0.2978	39	0.2965	40	0.2960	41	0.2969	42	0.2978
45	0.3222	46	0.3165	47	0.3096	48	0.3003	49	0.3095	50	0.3167	51	0.3287	52	0.3305
55	0.3372	56	0.3337	57	0.3339	58	0.3326	59	0.3326	60	0.3317	61	0.3388	62	0.3444
65	0.3456	66	0.3394	67	0.3333	68	0.3293	69	0.3256	70	0.3222	71	0.3200	72	0.3180
75	0.3101	76	0.3047	77	0.3008	78	0.2985	79	0.2958	80	0.2940	81	0.2935	82	0.2934
85	0.2911	86	0.2905	87	0.2891	88	0.2874	89	0.2861	90	0.2851	91	0.2840	92	0.2835
95	0.2824	96	0.2823	97	0.2813	98	0.2796	99	0.2782	100	0.2771	101	0.2754	102	0.2741
105	0.2723	106	0.2719	107	0.2711	108	0.2701	109	0.2688	110	0.2688	111	0.2669	112	0.2657
113												113	0.2646	114	0.2640

APPENDIX F

EXPERIMENTAL INTENSITIES FOR CHOCEB'S

CURVE I. WAVELENGTH=0.05025A SAMPLE DISTANCE=296.16MM															
7	0.2353	8	0.1621	9	0.1184	10	0.0974	11	0.0962	12	0.0999	13	0.1004	14	0.0974
17	0.1392	18	0.1556	19	0.1574	20	0.1506	21	0.1467	22	0.1439	23	0.1423	24	0.1377
27	0.1492	28	0.1577	29	0.1582	30	0.1525	31	0.1448	32	0.1446	33	0.1440	34	0.1479
37	0.1540	38	0.1538	39	0.1516	40	0.1480	41	0.1441	42	0.1429	43	0.1459	44	0.1504
45												45	0.1548		
CURVE II. WAVELENGTH=0.05025A SAMPLE DISTANCE=296.16MM															
7	0.2320	8	0.1570	9	0.1105	10	0.0935	11	0.0910	12	0.0948	13	0.0935	14	0.0967
17	0.1396	18	0.1552	19	0.1565	20	0.1498	21	0.1444	22	0.1432	23	0.1412	24	0.1367
27	0.1483	28	0.1558	29	0.1581	30	0.1514	31	0.1439	32	0.1413	33	0.1438	34	0.1474
37	0.1520	38	0.1524	39	0.1500	40	0.1454	41	0.1418	42	0.1406	43	0.1431	44	0.1466
45												45	0.1490		
CURVE III. WAVELENGTH=0.05025A SAMPLE DISTANCE=296.16MM															
7	0.2322	8	0.2785	9	0.2045	10	0.1747	11	0.1707	12	0.1767	13	0.1772	14	0.1731
17	0.2393	18	0.2648	19	0.2689	20	0.2597	21	0.2510	22	0.2482	23	0.2454	24	0.2384
27	0.2547	28	0.2684	29	0.2698	30	0.2608	31	0.2486	32	0.2448	33	0.2480	34	0.2538
37	0.2618	38	0.2625	39	0.2590	40	0.2521	41	0.2461	42	0.2442	43	0.2480	44	0.2541
45												45	0.2577		
CURVE IV. WAVELENGTH=0.05025A SAMPLE DISTANCE=296.16MM															
7	0.3394	8	0.2761	9	0.1995	10	0.1708	11	0.1681	12	0.1740	13	0.1748	14	0.1711
17	0.2401	18	0.2656	19	0.2691	20	0.2595	21	0.2504	22	0.2477	23	0.2450	24	0.2384
27	0.2550	28	0.2682	29	0.2698	30	0.2604	31	0.2490	32	0.2444	33	0.2477	34	0.2536
37	0.2610	38	0.2612	39	0.2576	40	0.2504	41	0.2438	42	0.2420	43	0.2461	44	0.2524
45												45	0.2558		
CURVE V. WAVELENGTH=0.05006A SAMPLE DISTANCE=95.45MM															
35	0.2249	36	0.2190	37	0.2153	38	0.2134	39	0.2114	40	0.2094	41	0.2097	42	0.2145
43	0.2498	44	0.2576	45	0.2626	46	0.2653	47	0.2683	48	0.2724	49	0.2765	50	0.2823
51	0.2958	52	0.2936	53	0.2879	54	0.2819	55	0.2789	56	0.2782	57	0.2782	58	0.2780
59	0.2702	60	0.2621	61	0.2566	62	0.2532	63	0.2507	64	0.2497	65	0.2501	66	0.2491
67	0.2395	68	0.2368	69	0.2354	70	0.2345	71	0.2331	72	0.2333	73	0.2329	74	0.2319
75	0.2272	76	0.2267	77	0.2266	78	0.2261	79	0.2260	80	0.2258	81	0.2258	82	0.2254
83	0.2255	84	0.2282	85	0.2295	86	0.2264	87	0.2265	88	0.2235	89	0.2237	90	0.2235
91	0.2237	92	0.2233	93	0.2233	94	0.2233	95	0.2233	96	0.2233	97	0.2233	98	0.2233
99	0.2233	100	0.2233	101	0.2233	102	0.2233	103	0.2233	104	0.2233	105	0.2233	106	0.2233
107	0.2233	108	0.2233	109	0.2233	110	0.2233	111	0.2233	112	0.2233	113	0.2233	114	0.2233
115	0.2233	116	0.2233	117	0.2233	118	0.2233	119	0.2233	120	0.2233	121	0.2233	122	0.2233
123	0.2233	124	0.2233	125	0.2233	126	0.2233	127	0.2233	128	0.2233	129	0.2233	130	0.2233
CURVE VI. WAVELENGTH=0.05006A SAMPLE DISTANCE=95.45MM															
35	0.2214	36	0.2155	37	0.2127	38	0.2105	39	0.2090	40	0.2073	41	0.2075	42	0.2122
43	0.2504	44	0.2593	45	0.2645	46	0.2673	47	0.2708	48	0.2742	49	0.2791	50	0.2855
51	0.2999	52	0.2967	53	0.2904	54	0.2853	55	0.2813	56	0.2801	57	0.2805	58	0.2802
59	0.2702	60	0.2641	61	0.2587	62	0.2546	63	0.2523	64	0.2516	65	0.2511	66	0.2498
67	0.2401	68	0.2366	69	0.2347	70	0.2340	71	0.2341	72	0.2344	73	0.2340	74	0.2337
75	0.2283	76	0.2280	77	0.2285	78	0.2304	79	0.2313	80	0.2314	81	0.2311	82	0.2298
83	0.2279	84	0.2286	85	0.2292	86	0.2295	87	0.2293	88	0.2293	89	0.2293	90	0.2293
91	0.2275	92	0.2279	93	0.2282	94	0.2284	95	0.2281	96	0.2275	97	0.2270	98	0.2270
99	0.2270	100	0.2271	101	0.2271	102	0.2271	103	0.2271	104	0.2271	105	0.2271	106	0.2271
107	0.2271	108	0.2271	109	0.2271	110	0.2271	111	0.2271	112	0.2271	113	0.2271	114	0.2271
115	0.2271	116	0.2271	117	0.2271	118	0.2271	119	0.2271	120	0.2271	121	0.2271	122	0.2271
123	0.2271	124	0.2271	125	0.2271	126	0.2271	127	0.2271	128	0.2271	129	0.2271	130	0.2271
CURVE VII. WAVELENGTH=0.05006A SAMPLE DISTANCE=95.45MM															
35	0.2284	36	0.2224	37	0.2186	38	0.2141	39	0.2139	40	0.2123	41	0.2122	42	0.2166
43	0.2542	44	0.2632	45	0.2679	46	0.2709	47	0.2744	48	0.2786	49	0.2803	50	0.2855
51	0.3011	52	0.2987	53	0.2926	54	0.2870	55	0.2838	56	0.2829	57	0.2833	58	0.2831
59	0.2734	60	0.2679	61	0.2622	62	0.2582	63	0.2558	64	0.2547	65	0.2544	66	0.2533
67	0.2428	68	0.2402	69	0.2380	70	0.2370	71	0.2374	72	0.2377	73	0.2370	74	0.2354
75	0.2338	76	0.2306	77	0.2314	78	0.2323	79	0.2322	80	0.2324	81	0.2322	82	0.2311
83	0.2296	84	0.2304	85	0.2304	86	0.2309	87	0.2307	88	0.2307	89	0.2315	90	0.2315
91	0.2286	92	0.2292	93	0.2294	94	0.2295	95	0.2295	96	0.2288	97	0.2285	98	0.2288
99	0.2285	100	0.2285	101	0.2285	102	0.2285	103	0.2285	104	0.2285	105	0.2285	106	0.2285
107	0.2285	108	0.2285	109	0.2285	110	0.2285	111	0.2285	112	0.2285	113	0.2285	114	0.2285
115	0.2285	116	0.2285	117	0.2285	118	0.2285	119	0.2285	120	0.2285	121	0.2285	122	0.2285
123	0.2285	124	0.2285	125	0.2285	126	0.2285	127	0.2285	128	0.2285	129	0.2285	130	0.2285
CURVE VIII. WAVELENGTH=0.05006A SAMPLE DISTANCE=95.45MM															
35	0.2193	36	0.2144	37	0.2108	38	0.2095	39	0.2078	40	0.2060	41	0.2062	42	0.2111
43	0.2485	44	0.2568	45	0.2614	46	0.2639	47	0.2676	48	0.2709	49	0.2757	50	0.2816
51	0.2933	52	0.2916	53	0.2854	54	0.2801	55	0.2770	56	0.2741	57	0.2757	58	0.2756
59	0.2654	60	0.2598	61	0.2543	62	0.2512	63	0.2490	64	0.2481	65	0.2489	66	0.2453
67	0.2362	68	0.2335	69	0.2325	70	0.2316	71	0.2299	72	0.2305	73	0.2302	74	0.2291
75	0.2246	76	0.2244	77	0.2250	78	0.2261	79	0.2265	80	0.2264	81	0.2254	82	0.2245
83	0.2226	84	0.2239	85	0.2233	86	0.2231	87	0.2238	88	0.2239	89	0.2239	90	0.2239
91	0.2229	92	0.2214	93	0.2211	94	0.2202	95	0.2201	96	0.2205	97	0.2197	98	0.2193
99	0.2206	100	0.2210	101	0.2210	102	0.2210	103	0.2210	104	0.2210	105	0.2210	106	0.2210
107	0.2210	108	0.2210	109	0.2210	110	0.2210	111	0.2210	112	0.2210	113	0.2210	114	0.2210
115	0.2210	116	0.2210	117	0.2210	118	0.2210	119	0.2210	120	0.2210	121	0.2210	122	0.2210
123	0.2210	124	0.2210	125	0.2210	126	0.2210	127	0.2210	128	0.2210	129	0.2210	130	0.2210

Appendix G

Correlation matrices¹ for $(\text{CH}_3)_2\text{GeH}_2$, $(\text{CH}_3)_4\text{Ge}$, CH_3GeF_3 and CH_3GeBr_3 $(\text{CH}_3)_2\text{GeH}_2$

	$r(\text{Ge-C})$	$r(\text{C-H})$	$r(\text{Ge-H})$	$\lambda(\text{Ge-C})$	$\lambda(\text{Ge-H})$
σ	0.001	0.0054	0.0115	0.0013	0.0049
$r(\text{Ge-C})$	1				
$r(\text{C-H})$	-0.22	1			
$r(\text{Ge-H})$	0.42	0.02	1		
$\lambda(\text{Ge-C})$	0.04	0.02	-0.08	1	
$\lambda(\text{Ge-H})$	-0.10	0.05	-0.07	-0.21	1

 $(\text{CH}_3)_4\text{Ge}$

	$r(\text{Ge-C})$	$r(\text{C-H})$	$\lambda(\text{Ge-C})$	$\lambda(\text{Ge-H})$	$\lambda(\text{C..C})$
σ	0.0016	0.0094	0.0023	0.0097	0.0098
$r(\text{Ge-C})$	1				
$r(\text{C-H})$	-0.20	1			
$\lambda(\text{Ge-C})$	0.06	0.15	1		
$\lambda(\text{Ge-H})$	-0.21	0.09	-0.18	1	
$\lambda(\text{C..C})$	0.01	-0.08	0.00	-0.13	1

Appendix G (continued)

CH ₃ GeF ₃					
	r(Ge-C)	r(C-H)	r(Ge-F)	<FGeC	
σ	0.0041	0.0258	0.0009	0.2660	λ(Ge-F) λ(C..F) λ(F..F)
r(Ge-C)	1				0.0009 0.0059 0.0053
r(C-H)	-0.07	1			
r(Ge-F)	0.29	-0.01	1		
<FGeC	-0.18	-0.06	0.06	1	
λ(Ge-F)	0.25	-0.03	0.14	-0.05	1
λ(C..F)	-0.06	-0.05	-0.03	0.23	-0.01 1
λ(F..F)	0.13	0.41	0.08	0.01	0.03 -0.12 1

CH ₃ GeBr ₃					
	r(Ge-C)	r(Ge-Br)	<BrCGe	λ(Ge-C)	λ(Ge-Br) λ(C..Br) λ(Br..Br)
σ	0.0113	0.0006	0.1002	0.0189	0.0006 0.0121 0.0020
r(Ge-C)	1				
r(Ge-Br)	-0.04	1			
<BrCGe	-0.04	0.35	1		
λ(Ge-C)	0.09	-0.03	0.01	1	
λ(Ge-Br)	0.01	0.08	0.04	-0.02	1

Appendix G (continued)

$\ell(C..Br)$	-0.06	0.00	-0.02	0.02	-0.01	1
$\ell(Br..Br)$	0.07	0.02	-0.03	0.04	0.01	0.37
						1

1 The calculated least squares standard deviation σ is given by

$$\sigma = \left[\frac{B_{11}^{-1} V' W W}{n - m} \right]^{1/2}$$

and the correlation coefficients by

$$\rho_{1j} = \frac{B_{1j}^{-1}}{\left(\frac{B_{11}^{-1} B_{jj}^{-1}}{B_{jj}^{-1}} \right)^{1/2}}$$

APPENDIX H

EXPERIMENTAL INTENSITIES FOR ACETIC 4,6-DIMETHYL TRIMETHYLENE SULFITE

DATA SET I.

WAVELENGTH=0.04871A SAMPLE DISTANCE=262.97MM

14	0.4659	15	0.5687	16	0.6668	17	0.7210	18	0.7129	19	0.6665	20	0.6036	21	0.5542
22	0.5258	23	0.5314	24	0.5703	25	0.6299	26	0.6856	27	0.7151	28	0.7180	29	0.7119
30	0.7091	31	0.7124	32	0.7155	33	0.7128	34	0.7046	35	0.6945	36	0.6922	37	0.6994
38	0.7170	39	0.7465	40	0.7798	41	0.8088	42	0.8271	43	0.8275	44	0.8139	45	0.7988
46	0.7922	47	0.7969	48	0.8053	49	0.8147	50	0.8229	51	0.8328	52	0.8456	53	0.8574
54	0.8651	55	0.8754	56	0.8883	57	0.8998	58	0.9060	59	0.9107	60	0.9155	61	0.9215
62	0.9290														

WAVELENGTH=0.04871A SAMPLE DISTANCE=262.97MM

40	0.5145	41	0.5373	42	0.5520	43	0.5543	44	0.5444	45	0.5318	46	0.5231	47	0.5240
48	0.5294	49	0.5331	50	0.5366	51	0.5414	52	0.5487	53	0.5558	54	0.5620	55	0.5671
56	0.5729	57	0.5783	58	0.5816	59	0.5819	60	0.5796	61	0.5793	62	0.5814	63	0.5845
64	0.5874	65	0.5909	66	0.5963	67	0.6028	68	0.6095	69	0.6144	70	0.6186	71	0.6233
72	0.6288	73	0.6352	74	0.6412	75	0.6468	76	0.6520	77	0.6568	78	0.6619	79	0.6659
80	0.6699	81	0.6754	82	0.6823	83	0.6904	84	0.6996	85	0.7078	86	0.7157	87	0.7229
88	0.7300	89	0.7380	90	0.7452	91	0.7523	92	0.7588	93	0.7670	94	0.7748	95	0.7830
96	0.7916														

DATA SET II.

WAVELENGTH=0.04871A SAMPLE DISTANCE=167.42MM

6	0.1629	7	0.1417	8	0.1322	9	0.1254	10	0.1220	11	0.1133	12	0.1086	13	0.1136
14	0.1315	15	0.1557	16	0.1793	17	0.1915	18	0.1879	19	0.1771	20	0.1609	21	0.1501
22	0.1444	23	0.1467	24	0.1571	25	0.1729	26	0.1866	27	0.1927	28	0.1932	29	0.1915
30	0.1913	31	0.1921	32	0.1926	33	0.1918	34	0.1895	35	0.1870	36	0.1871	37	0.1906
38	0.1961	39	0.2050	40	0.2144	41	0.2223	42	0.2266	43	0.2271	44	0.2237	45	0.2206
46	0.2206	47	0.2230	48	0.2261	49	0.2288	50	0.2314	51	0.2352	52	0.2392	53	0.2428
54	0.2459	55	0.2493	56	0.2532	57	0.2570	58	0.2593	59	0.2603	60	0.2613	61	0.2633
62	0.2656	63	0.2674												

WAVELENGTH=0.04871A SAMPLE DISTANCE=167.42MM

40	0.2064	41	0.2165	42	0.2224	43	0.2243	44	0.2209	45	0.2152	46	0.2127	47	0.2136
48	0.2161	49	0.2177	50	0.2190	51	0.2216	52	0.2250	53	0.2287	54	0.2326	55	0.2352
56	0.2380	57	0.2412	58	0.2431	59	0.2437	60	0.2440	61	0.2442	62	0.2453	63	0.2466
64	0.2481	65	0.2495	66	0.2518	67	0.2545	68	0.2578	69	0.2599	70	0.2619	71	0.2640
72	0.2670	73	0.2686	74	0.2720	75	0.2743	76	0.2769	77	0.2788	78	0.2814	79	0.2835
80	0.2855	81	0.2879	82	0.2914	83	0.2949	84	0.2987	85	0.3025	86	0.3060	87	0.3101
88	0.3133	89	0.3169	90	0.3208	91	0.3250	92	0.3276	93	0.3310				

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